

New algorithm for efficient Bloch-waves calculations of orientation-sensitive ELNES

Ján Ruzs,^{1,2} Shunsuke Muto,³ and Kazuyoshi Tatsumi³

¹*Department of Physics, Uppsala University, Box 530, S-751 21 Uppsala, Sweden*

²*Institute of Physics, Academy of Sciences of the Czech Republic,
Na Slovance 2, CZ-182 21 Prague, Czech Republic*

³*Department of Materials, Physics, and Energy Engineering,
Nagoya University, Chikusa, Nagoya 464-8603, Japan*

We discuss in detail the Bloch waves method for calculation of energy and orientation dependent scattering cross-section for inelastic scattering of electrons on crystals. Convergence properties are investigated and a new algorithm with superior timing and accuracy is described. The new method is applied to calculations of intensity of weakly excited spots, maps of magnetic signal, and tilt series from zone axis orientation towards three-beam orientation.

Keywords: transmission electron microscopy, density functional theory, dynamical diffraction theory, Bloch waves, electron magnetic circular dichroism

I. INTRODUCTION

Dynamical diffraction theory describes the multiple elastic scattering of electrons through a crystal. The theory has been formulated almost a century ago, originally for photons by Bragg¹ and von Laue². Later it was extended also to electrons on the basis of wave-particle duality, as proven by the famous experiments of Davisson and Germer³, and Thomson and Reid⁴. The importance of inelastic electron scattering and the wealth of information contained in this process was for the first time observed by Kikuchi⁵, who observed a complicated pattern of lines going beyond the diffraction patterns expected for elastic scattering of light or electrons. The diffraction theory that included quantitative description of Kikuchi patterns was given by Kainuma following qualitative theories of Shinohara and von Laue⁶. Various sources of energy loss, such as excitations of phonons or valence electrons to conduction band or core electrons to conduction band were discussed by Okamoto et al.⁷

The inelastic electron scattering is typically described as a three-step process: 1) an elastic scattering of the probe electron from entrance surface to a selected atom in the sample, 2) an inelastic event exchanging the momentum and energy between a probe and sample electron, and 3) an elastic propagation of the scattered probe electron towards the exit sample surface. This needs to be summed over all possible inelastic scattering centers. Such description holds, when there is only one inelastic scattering event for each observed probe electron. If its mean free path is much longer than the thickness of the sample, then it is a good approximation. Otherwise one needs to consider multiple inelastic scattering⁸.

The inelastic event is described by mixed dynamic form factor (MDFF) introduced by Kohl and Rose¹⁰. For calculations of MDFF, there are various levels of sophistication, ranging from an isotropic dipole approximation $\mathbf{q} \cdot \mathbf{q}'$, through a parametrized atomic multiplets description¹³, configuration interaction¹⁴, to a density functional theory (DFT) evaluation^{9,12}. Recently a DFT-based dipole model for MDFFs has been proposed using

local electronic structure properties to set the coefficients in the dipole approximation¹¹, thereby providing a realistic DFT-based MDFF model with efficiency equal to a simple dipole approximation (at the cost of losing the fine structure in energy dependence).

The elastic scattering can be simulated by one of the two major methods widely used today, the multislice method^{15,16} and Bloch-waves (BW) method^{17,18}. The latter one is typically more efficient for periodic structures without defects, while the multislice method has an advantage when dealing with large non-periodic structures or structures with defects. In this article we will study the convergence properties of the BW method in electron energy-loss near edge structure (ELNES) calculations and describe an efficient algorithm for BW summation.

II. BLOCH-WAVES THEORY OF ELNES

In Ref. 9 we have described our theoretical approach to simulate general orientation-sensitive ELNES experiment from first principles. We present here a brief summary of the key equations.

The first step is a solution of the secular equation for a fast electron of known energy moving in and out of the crystal. Expanding the solutions - Bloch waves - into plane waves (indexed by reciprocal lattice vectors \mathbf{g}, \mathbf{h}) the secular equation has the following form:

$$\sum_{\mathbf{g}} \left[\left(K^2 - (\mathbf{k}^{(j)} + \mathbf{g})^2 \right) + \sum_{\mathbf{h} \neq 0} U_{\mathbf{h}} C_{\mathbf{g}-\mathbf{h}}^{(j)} \right] e^{i(\mathbf{k}^{(j)} + \mathbf{g}) \cdot \mathbf{r}} = 0 \quad (1)$$

where $K^2 = U_0 + 2meE/\hbar^2$, m and e are, respectively, the electron mass and charge, $U_{\mathbf{h}} = 2meV_{\mathbf{h}}/\hbar^2$ where $V_{\mathbf{h}}$ are the Fourier components of the crystal potential (including the $\mathbf{h} = 0$ case in the definition of K^2), E is acceleration voltage, and $\mathbf{k}^{(j)} = \mathbf{k} + \gamma^{(j)}\mathbf{n}$ are Bloch vectors related to the beam direction \mathbf{k} and sample surface normal \mathbf{n} .

Solution of the secular equation is a set of Bloch waves

indexed by j , given by so called Bloch coefficients $C_{\mathbf{g}}^{(j)}$ and elongations of the wave vector $\gamma^{(j)}$.

Evaluation of the double differential scattering cross-section involves calculation of the following sum [see, e.g., Ref. 9, Eq. (24)]

$$\begin{aligned} \frac{\partial^2 \sigma}{\partial \Omega \partial E} &= \sum_{\mathbf{g}\mathbf{h}\mathbf{g}'\mathbf{h}'} \frac{1}{N_{\mathbf{u}}} \sum_{\mathbf{u}} \frac{S_{\mathbf{u}}(\mathbf{q}, \mathbf{q}', E)}{q^2 q'^2} e^{i(\mathbf{q}-\mathbf{q}')\cdot\mathbf{u}} \\ &\times \sum_{jlj'l'} Y_{\mathbf{g}\mathbf{h}\mathbf{g}'\mathbf{h}'}^{jlj'l'} T_{jlj'l'}(t) \end{aligned} \quad (2)$$

where

$$\begin{aligned} Y_{\mathbf{g}\mathbf{h}\mathbf{g}'\mathbf{h}'}^{jlj'l'} &= C_{\mathbf{0}}^{(j)*} C_{\mathbf{g}}^{(j)} D_{\mathbf{0}}^{(l)} D_{\mathbf{h}}^{(l)*} \\ &\times C_{\mathbf{0}}^{(j')} C_{\mathbf{g}'}^{(j')*} D_{\mathbf{0}}^{(l')} D_{\mathbf{h}'}^{(l')*}. \end{aligned} \quad (3)$$

Here $C_{\mathbf{g}}^{(j)}$ and $D_{\mathbf{h}}^{(l)}$ are the Bloch coefficients for the incoming and outgoing Bloch fields. The quantity $\frac{S_{\mathbf{u}}(\mathbf{q}, \mathbf{q}', E)}{q^2 q'^2}$ is the mixed-dynamic form-factor (MDFF)^{10,12} of atom with basis vector \mathbf{u} , divided by squares of momentum transfer vectors \mathbf{q}, \mathbf{q}' originating from the Fourier transform of Coulomb interaction between electrons. $T_{jlj'l'}(t)$ is a thickness function, which depends on Bloch wave indices and experimental geometry. $N_{\mathbf{u}}$ is a number of atoms in the unit cell, where \mathbf{u} is a base vector. Momentum transfer vectors actually depend on several indices, but to simplify the notation we will not write this dependence explicitly. For evaluation of MDFFs we typically use an approximation

$$\mathbf{q} = \mathbf{k}_{\text{out}}^{(l)} - \mathbf{k}_{\text{in}}^{(j)} + \mathbf{h} - \mathbf{g} \approx \mathbf{k}_{\text{out}} - \mathbf{k}_{\text{in}} + \mathbf{h} - \mathbf{g} \quad (4)$$

which neglects elongations of the wave vectors $\gamma^{(j)}, \gamma^{(l)}$.

III. SUMMATION ALGORITHMS

Here we will discuss two known algorithms used in Bloch wave calculations and propose new algorithms for improved convergence and efficiency of summation.

A. Manual selection of beams

The simplest one is based on a choice of beams “by hand” let’s say on a base of visible spots in diffraction pattern, or all beams from a set of integer indices below certain cut-off, or beams on a systematic row, etc. For example, in^{9,20} we used a systematic row approximation both for secular equation and summation, i.e., we picked only a set of ~ 10 beams along the systematic row of reflections. Pragmatic hand-selection of beams is widespread in literature, see for example Refs.^{21–25}.

B. Excitation error and extinction distance based selection of beams

In a previous work⁹ it was suggested to choose beams on the base of their excitation error $s_{\mathbf{g}}$ and extinction distances $\xi_{\mathbf{g}}$. Their product forms a dimensionless variable $w_{\mathbf{g}}$, for which one sets a cut-off criterion. This method picks beams that follow the Ewald sphere and provides automatically a more economic and accurate description. A variant of this method uses two different cut-offs for the $w_{\mathbf{g}}$ - one for the secular equation (hundreds or thousands of beams) and one for the summation (10–15 beams). We can safely take a rather large set of beams in the first step – the solution of the secular equation – and then for the summation we pick only a subset of the beams and Bloch waves. The selection of the subset of Bloch waves is based on their norm within the subspace of beams selected for summation⁹. This has been applied in several publications.^{19,20,26,27}

Both this approach and the manual selection of beams identify a set of beams and Bloch waves and then they count all the cross-terms. That is, if we have $N_{\mathbf{g}}$ beams and N_j Bloch waves, the summation runs over $N_{\mathbf{g}}^4 N_j^4$ terms, which can be already a huge number for only 10 \mathbf{g} -vectors and 10 Bloch waves. A detailed inspection of these terms shows, that majority of them are actually negligible. That means summation of many negligible terms, which is not only inefficient but also contributes to the propagation of machine rounding off errors.

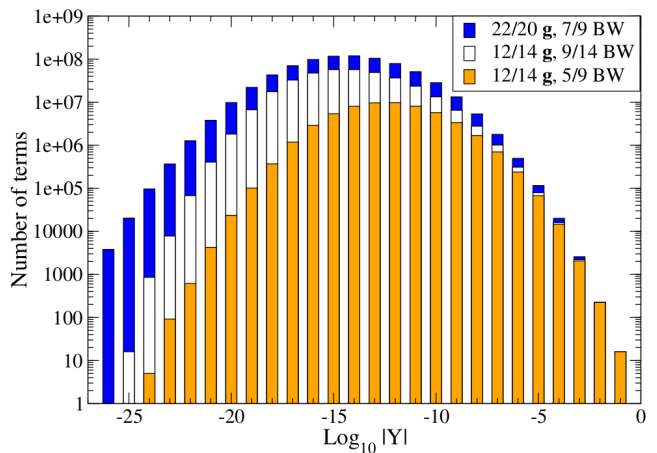


FIG. 1. Histogram of the distribution of $Y_{\mathbf{g}\mathbf{h}\mathbf{g}'\mathbf{h}'}^{jlj'l'}$ terms magnitudes as a function of number of \mathbf{g} -vectors and Bloch waves (BW).

To illustrate the finding, we have set up test calculations for bcc iron at 300 keV acceleration voltage, incoming beam direction [016] resulting in a 3-beam orientation with $\mathbf{G} = (200)$, Laue circle center at $(0, 0)$, and detector orientation $(\frac{G}{2}, \frac{G}{2})$. By setting the maximum $w_{\mathbf{g}} = 10^5$ for the secular equation, the program identified a list of 628 and 627 \mathbf{g} -vectors for incoming and outgoing beams, respectively. For the summation we set the maximum

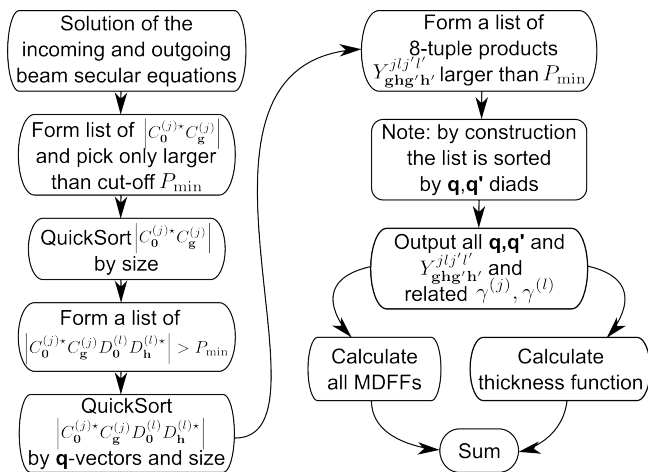


FIG. 2. Flowchart of the algorithm for automatic selection of dominant terms.

$w_{\mathbf{g}} = 500$, which filtered the list down to 12 and 14 \mathbf{g} -vectors. These contained (000), $\pm(200)$, $\pm(110)$, $\pm(1\bar{1}0)$, (310), $(\bar{3}10)$, $\pm(400)$ and (020). The outgoing beam included on top of that (020) and (510). With the criterion of minimal norm of the Bloch wave in the subspace of identified beams 0.01 we got 5 and 9 Bloch waves, while for criterion 0.005 we got 8 and 14 Bloch waves for incoming and outgoing beam, respectively. In total that makes above 57 million vs 354 million terms. The computing time of one thickness profile, that is 100 thickness values at a fixed energy loss 708 eV, was 45s vs 258s, respectively, performed on a single 2.0 GHz Intel Pentium 4 Xeon CPU.

Instead of increasing number of Bloch waves, we also tested the effect of enlarging the set of \mathbf{g} -vectors in summation by setting $w_{\mathbf{g}} = 800$. The number of selected \mathbf{g} -vectors grows to 22 and 20, and some of the hkl beams have nonzero l (higher order Laue zones). Keeping minimal norm for Bloch waves 0.01 we got 7 and 9 Bloch waves. In total it gives 768 millions of terms, which were summed in 1308s.

Figure 1 shows histograms of distributions of the sizes of all these terms. If, for example, all of the 1000 dominant terms (for our fixed selection of beams) are included in the summation, we are also including into the sum many millions of terms with much smaller magnitudes, majority of them having negligible influence on results. As we will show below, 1000 dominant terms might not always be enough in terms of convergence. Then one can easily conclude that requiring a summation over, e.g., 10^5 dominant terms would require such a number of beams and Bloch waves that we would end up summing billions of terms, greatly wasting computational resources.

C. Automatic selection of dominant terms

The new methods described below are designed to 1) avoid summation of negligible terms, 2) improve the scaling of the summation. They are built on top of the $w_{\mathbf{g}}$ based selection of the beams for diagonalization, assuming that we take a sufficiently large set of beams for the secular equation, typically hundreds up to few thousands ($N_{\mathbf{g}} \sim 100$ —5000). The core of our new method is an algorithm for selection of dominant terms out of all $N_{\mathbf{g}}^8$ terms that are generated by a secular equation with $N_{\mathbf{g}}$ beams.

The algorithm is controlled by a single cut-off criterion P_{\min} . In the first step we create lists of diads of the Bloch coefficients, $|C_0^{(j)} C_{\mathbf{g}}^{(j)}| > P_{\min}$ and $|D_0^{(l)} D_{\mathbf{h}}^{(l)}| > P_{\min}$, for incoming and outgoing beam, respectively. Since none of $|C_{\mathbf{g}}^{(j)}|$ or $|D_{\mathbf{h}}^{(l)}|$ is larger than one, we can safely ignore all Bloch waves, for which $|C_0^{(j)}| < P_{\min}$ or $|D_0^{(l)}| < P_{\min}$. The resulting array is sorted according to decreasing magnitude using QuickSort algorithm. In total, this step has computational complexity $O(N_{\mathbf{g}}^2 + N_2 \log N_2)$, where N_2 is length of the list of diads.

The next step is creation of a list of quadruple products $|C_0^{(j)} C_{\mathbf{g}}^{(j)} D_0^{(l)} D_{\mathbf{h}}^{(l)}| > P_{\min}$. This operation is $O(N_4)$, where N_4 is the number of the quadruples larger than P_{\min} , because the lists of diads were sorted by magnitude. The maximum number of failed comparisons is $O(N_2)$, where N_2 is the length of the longer of the two lists of diads formed in previous step.

The list of quadruples is again sorted with QuickSort algorithm, but this time first by the \mathbf{q} -vector given by $\mathbf{h} - \mathbf{g}$ and then by magnitude. This operation costs $O(N_4 \log N_4)$ operations. Now we have prepared data for the final step, which is an identification and output of octuple products larger than P_{\min} and, simultaneously, output of the \mathbf{q}, \mathbf{q}' diads for the MDFF calculation. Benefiting from the way how we sorted the list of quadruples, we can serially process the list and output \mathbf{q}, \mathbf{q}' and corresponding octuples without actually holding the array of octuples in memory. The number of operations is $O(N_8)$, where N_8 is the number of octuples larger than P_{\min} . Maximum number of failed comparisons is given by the number of \mathbf{q}, \mathbf{q}' diads, which is well below N_8 .

As a graphical summary of the main steps of the algorithm, a schematic flowchart diagram has been plotted in Fig. 2.

Memory requirements are very favorable. Most often, the largest arrays are the $N_{\mathbf{g}} \times N_{\mathbf{g}}$ matrices used in the secular equation. The lists of quadruples rarely reach this lengths, only perhaps for extremely small P_{\min} of the order below 10^{-6} . The octuples are never held in memory, since the array is created and output serially based on the favorable sorting of the list of quadruples.

As will be shown below, N_8 can be orders of magnitude below the $N_{\mathbf{g}}^4 N_j^4$, even if naturally it has to be proportional to that. Its value strongly depends on P_{\min} , and it turns out to be approximately inverse proportional to

P_{\min} , i.e., $N_8 \propto P_{\min}^{-1}$. Tuning the P_{\min} allows to find a suitable compromise between the speed and accuracy, which will be discussed below. Also note that we are selecting largest terms from a large set of beams, of the order of 1000, which may not be feasible to treat by above-mentioned methods. Therefore it can happen that stricter selections of beams and including all cross-terms can lead to less accurate calculations because some singular contributions of considerable magnitude beams can be missed. Examples of such situations will be shown below.

D. Automatic term selection with consideration of MDFF asymptotical behavior

A modification of the algorithm is possible if we take an advantage of the dipole-type asymptotic behavior of the MDFFs and the $1/q^2q'^2$ denominator. For large \mathbf{q} -vectors the denominator suppresses the terms, therefore we can do an even more efficient rejection of the negligible terms.

In the dipole approximation, MDFF is proportional to

$$S(\mathbf{q}, \mathbf{q}', E) \propto \mathbf{q} \cdot \bar{N}(E) \cdot \mathbf{q}' + (\mathbf{q} \times \mathbf{q}') \cdot \mathbf{M}(E) \quad (5)$$

where \bar{N} is a energy-resolved tensor dependent on density of states, local anisotropies and spin-orbit coupling and \mathbf{M} is an energy-resolved vector function of local magnetic properties¹¹. Ignoring the energy dependence, the asymptotic behavior of MDFF as a function of \mathbf{q}, \mathbf{q}' vectors is $S(\mathbf{q}, \mathbf{q}') \propto qq'$. Combining this with the denominator $1/q^2q'^2$ accompanying every MDFF, we obtain $1/qq'$ asymptotic behavior of terms. Note that the dipole behavior of MDFF is valid only for small \mathbf{q}, \mathbf{q}' . For large ones, MDFF goes to zero instead of increasing boundlessly. In this sense, the $1/qq'$ asymptotic behavior can be considered as an upper bound.

In order to keep a dimensionless variable for the cut-off criterion, we attach a factor $q_0/q_{\mathbf{gh}}$ to the lists of quadruples, where $\mathbf{q}_0 = \mathbf{k}_f - \mathbf{k}_i$ and $\mathbf{q}_{\mathbf{gh}} = \mathbf{q}_0 + \mathbf{h} - \mathbf{g}$. A small complication arises from this choice, since it is possible that the ratio $q_0/q_{\mathbf{gh}}$ is larger than one for some larger scattering angles or energy losses. For example, when $\mathbf{q}_0 = \mathbf{k}_f - \mathbf{k}_i \approx \mathbf{G}$ for some allowed reflection \mathbf{G} , then $|\mathbf{q}_0| > |\mathbf{q}_0 - \mathbf{G}|$. Then in some of the summed terms the $\mathbf{q}_{\mathbf{gh}}$ can be equal to $\mathbf{q}_0 - \mathbf{G}$, and in such case we get $q_0/q_{\mathbf{gh}} > 1$. For such eventualities we need to make sure that our lists of diads have a ‘reserve’, therefore the cut-off for the list of diads needs to be reduced. We have implemented a cut-off $P_{\min}/10$ for the list of diads, which is an arbitrary choice, yet it turned out to be both safe and not too costly when compared to other list operations in the algorithm. The rest of algorithm is unchanged, only when outputting the list of selected octuples, we remove the asymptotic factor $q_0^2/q_{\mathbf{gh}}q_{\mathbf{g}'\mathbf{h}'}$.

In the rest of the article, we use this modified automatic term selection algorithm (MATS).

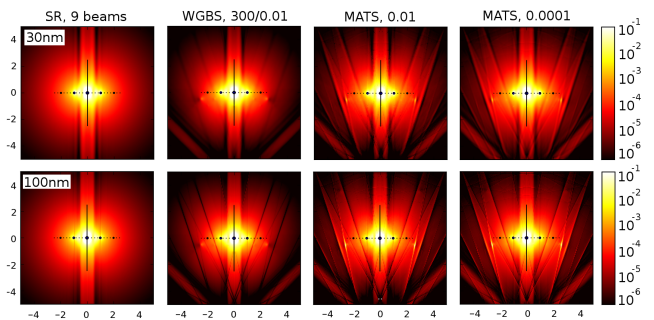


FIG. 3. Diffraction pattern of iron in 3-beam orientation with $\mathbf{G} = (200)$. Top row corresponds to 30nm sample thickness, bottom row is at 100nm. From left to right various methods of calculation have been used, namely systematic row (SR) approximation, $w_{\mathbf{g}}$ -based beam selection (WGBS) and automatic term selection with MDFF asymptotics (MATS) with two different cut-offs (see text for details). Intensities are on a logarithmic scale. The q_x, q_y axes are in multiples of G , where $\mathbf{G} = (200)$. Calculations were performed on a mesh of 201×201 pixels.

IV. RESULTS

In this section we compare the various methods of performing the Bloch waves summation, and demonstrate some of new possibilities offered by the MATS.

A. Weakly excited spots

In systematic row geometries, many simulations have been performed by only choosing beams along the systematic row. However, depending on the tilt of the beam, some of the spots outside the systematic row can be weakly excited. That can be easily missed in the systematic row approximation (SRA). Here we will show a simulation of a three-beam orientation for bcc iron with systematic row index $\mathbf{G} = (200)$ at acceleration voltage 300keV. We will consider a beam tilt of approximately 5 degrees from $[001]$ zone axis orientation, which corresponds approximately to incoming beam orientation $[0, 1, 10]$. In total 201×201 pixels were calculated in the range from $-5G$ to $+5G$ for both q_x and q_y .

Let’s compare precision and timing of SRA, $w_{\mathbf{g}}$ -based beam selection (WGBS) and the MATS. In the SRA we included beams up to $\pm 4\mathbf{G}$, in total 9 beams and the summation was performed over all 9 resulting Bloch waves. In the WGBS, for the secular equation we set the $w_{\mathbf{g}}$ cut-off to 100000, which resulted in approximately 630 beams. For the summation, we used cut-off 300, which was fulfilled by 8–14 beams. For the Bloch waves we required the subspace norm⁹ to be larger than 0.01, which was fulfilled by 4–27 Bloch waves out of the total number of approximately 630 Bloch waves. Finally, in the MATS we tested for summation the following two cut-off criteria: $P_{\min} = 0.01$ and 0.0001 .

TABLE I. Computational times (CPU) for the methods applied in Fig. 3. The third column summarizes statistical information about the minimum / average / maximum number of summation terms from the 201×201 calculated pixels.

Method	CPU [h]	Min/aver./max no. of terms
SRA, 9 beams	561	constant, $9^8 = 4.30 \times 10^7$
WGBS, 300/0.01	2800	$0.11/1.05/6.31 \times 10^8$
MATS, $P_{\min} = 10^{-2}$	43	$0.37/2.90/45.4 \times 10^2$
MATS, $P_{\min} = 10^{-4}$	98	$0.18/7.05/49.7 \times 10^4$

The resulting diffraction patterns are summarized in Fig. 3. The SRA calculation (1st column) shows the Bragg spots and a Kikuchi band perpendicular to the systematic row direction. Otherwise, no other features can be seen except for a gradual decrease of the scattering intensity towards larger scattering angles. When using the WGBS method, the resulting diffraction patterns (2nd column) show additional features, such as Kikuchi bands in lower quadrants and some spot-like features around $(q_x, q_y) = (\pm 5, -1)$. In the 3rd column there are results of the new summation method MATS picking only terms larger than 0.01. This calculation shows even more features, particularly the much better resolved Kikuchi bands passing from top corners of the diffraction pattern and crossing at the zone axis point (down, outside the range) at about 10 degrees tilt from the transmitted beam direction. In the last column we show a result of the MATS algorithm considering all terms larger than 0.0001. Visually it is not easy to spot differences from a calculation with cut-off 0.01, which demonstrates that already such a low cut-off provides reasonably converged diffraction pattern.

Regarding the computing costs, analysis is summarized in Table I. The fastest was the MATS simulation with $P_{\min} = 0.01$, which finished in 43 CPU hours. In this calculation, a large part of the time was spent in diagonalization of the secular matrices of dimension 630. This is clearly seen on MATS calculation with $P_{\min} = 0.0001$ —although it required on average summation of 200 times more terms, the computing time increased only by factor of two to 98 CPU hours. The SRA calculation took considerably more time, despite being much less accurate: the 9-beam calculation finished in 561 CPU hours. Note that here we are diagonalizing very small matrices, only 9-by-9, therefore practically all the computing time is spent in summation. Finally, the WGBS calculation required 2800 CPU hours, almost 70-times more than MATS with $P_{\min} = 0.01$, yet being of lower accuracy. While in our SRA calculation we always sum $9^8 = 43 \times 10^6$ terms, in MATS with $P_{\min} = 0.0001$ it was on average only 70×10^3 , in maximum reaching half million.

The most finding from these calculations is that MATS calculations provide substantially more detailed calculations and yet they require an order of magnitude less time than even a simple systematic row approximation with 9 beams.

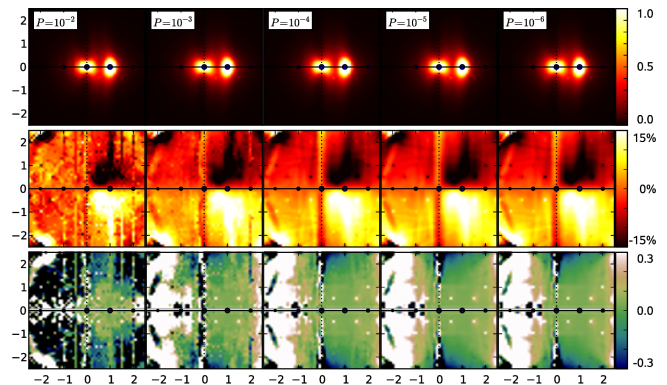


FIG. 4. Diffraction patterns (top row), relative up-down difference maps (middle row) and apparent m_L/m_S ratio maps (bottom row) a function of convergence parameter P_{\min} . Calculations were performed for 20nm layer of bcc iron in two-beam geometry with $\mathbf{G} = (200)$, at 300keV. The q_x, q_y axes are in multiples of G , where $\mathbf{G} = (200)$. Calculations were performed on a mesh of 51×51 pixels.

B. EMCD and m_L/m_S maps of iron

Electron magnetic circular dichroism (EMCD) is a recently developed experimental technique²⁸, which uses ELNES to extract the atom-specific magnetic characteristics, such as spin and orbital moments. For quantitative analyses it used sum rules^{29,30}, with which one can extract the ratio of orbital and spin moment of the atom of interest²⁷. These properties are highly sensitive functions of the edge-dependent ELNES spectra. As in experiment, also in simulations there is a high demand for precision. Fig. 4 shows the performance of MATS for the similar setup as above. A bcc iron sample was assumed oriented in a two-beam case with $\mathbf{G} = (200)$ and beam tilt of 10 degrees from $[001]$ zone axis, i.e., beam direction approximately along $[01\bar{6}]$. The acceleration voltage was set to 200 keV. The Fig. 4 shows diffraction pattern in the top row, distribution of the magnetic signal in the middle row and the map of the m_L/m_S ratio in the bottom row, as a function of the cut-off variable P_{\min} . The maps span an area of momentum transfers q_x, q_y from $-2.5G$ to $+2.5G$ on a mesh of 51×51 pixels.

The distribution of the magnetic signal is obtained as a difference of the diffraction pattern and its mirror image with respect to the systematic row mirror axis. In the figure, it is shown as a relative quantity, that means it is divided by the sum of the diffraction pattern and its mirror image. In other words, it is an antisymmetric part of the diffraction pattern divided by its symmetric part. We will not dwell here into details of spectrum processing on EMCD such as the continuum signal extraction or post-edge normalization—an interested reader can find information about these topics in³¹—since our main focus is the convergence of the Bloch waves calculation.

The maps of the m_L/m_S ratio are evaluated pixel-by-pixel from the magnetic signals M_{L_3} and M_{L_2} by the

TABLE II. Average lengths of double $\langle N_2 \rangle$, quadruple $\langle N_4 \rangle$ and octuple $\langle N_8 \rangle$ product lists, average number of momentum transfer diads per energy step $\langle N_{\mathbf{q}\mathbf{q}'} \rangle$ and computing times (total and per-pixel average) for maps in Fig. 4 as a function of convergence parameter P_{\min} . Times refer to a single Intel Pentium 4 Xeon processor at 2.5GHz.

P_{\min}	$\langle N_2 \rangle$	$\langle N_4 \rangle$	$\langle N_8 \rangle$	$\langle N_{\mathbf{q}\mathbf{q}'} \rangle$	t_{total}	$\langle t_1 \rangle$
10^{-2}	230	72	330	12	2h 16min	3.1s
10^{-3}	1355	750	4710	110	3h 18min	4.6s
10^{-4}	5545	5865	56830	705	11h 13min	15.5s
10^{-5}	23060	39760	570250	3600	83h 35min	115s
10^{-6}	73591	259739	5398841	16860	628h 14min	870s

following formula^{29,30}

$$\frac{m_L}{m_S} = \frac{2 M_{L_3} + M_{L_2}}{3 M_{L_3} - 2 M_{L_2}} \quad (6)$$

The M_{L_3} and M_{L_2} were integrated over the L_3 and L_2 ionization edge regions, respectively. More precisely, the energy range was set to 1.5–2.0eV above the Fermi level, containing 11 energy values. This range covers the peak of the unoccupied $3d$ spin-down states and it is the dominant contribution to the m_L and m_S measured by the sum rules²⁹. The limited energy range was chosen for the purpose of saving computational resources, since the DFT evaluation of MDFFs is rather costly and the focus here is on the BW part of the calculation. Nevertheless, it was tested that this energy range provides results representative of the whole energy range.

Intuitively, m_L/m_S should be a constant function throughout the diffraction plane. However, in fact, large variations occur due to asymmetries discussed in detail in³². The map of the m_L/m_S ratio is a highly sensitive function of the scattering cross-section and is an excellent test of the convergence properties for the MATS.

We have used the same $w_{\mathbf{g}}$ cut-off as in sub-section IV A, but we have varied the P_{\min} from 10^{-2} down to 10^{-6} and recorded some statistic information about the number of terms included in the summation—see Table II. Note that, as anticipated, the number of summed terms is inversely proportional to P_{\min} . Importantly, even at the most accurate calculation, the number of summed terms stays many orders of magnitude below 700^8 , demonstrating the high efficiency of the selection of terms by MATS.

Note that the diffraction pattern appears to be reasonably converged already for $P_{\min} = 10^{-2}$. An attentive reader might spot slight fuzziness, but nevertheless the differences between all five diffraction patterns are visually negligible. The relative difference map requires better convergence, at least $P_{\min} = 10^{-3}$, or better $P_{\min} = 10^{-4}$. Note the numerical noise at $P_{\min} = 10^{-3}$, particularly along the vertical line going through the transmitted beam. Although slight fuzziness remains at $P_{\min} = 10^{-3}$, but the accuracy is already satisfactory.

The most sensitive quantity presented here is the apparent m_L/m_S ratio. Since the orbital momentum in

iron is small, the numerator in the sum rule expression is a difference of two (typically small) differences of spectra, integrated over L_2 or L_3 edge, respectively. It requires high accuracy to obtain well converged maps. Calculation with $P_{\min} = 10^{-4}$ seems to produce reasonably converged results. The only visible difference at $P_{\min} = 10^{-5}$, when compared to $P_{\min} = 10^{-4}$, appears around $(-1, \pm 1)G$ and $(2, \pm 2)G$ positions. The results of calculation with $P_{\min} = 10^{-6}$ are visually indistinguishable from $P_{\min} = 10^{-5}$.

Looking back into the Table II, we see that, for most of our purposes, $P_{\min} = 10^{-4}$ provides highly converged results at average costs 3-times lower than the simplest testing calculation discussed in Sec. III B. Nevertheless, this criterion can depend on the crystal structure and orientation of interest and always should be tested, when performing simulations of new systems.

C. Tilting crystal from the zone axis orientation

High efficiency of the MATS method allows us to approach experimental geometries, which require a large number of beams for a converged simulation. Particularly, the zone axis orientation is highly computationally intensive. Here we will follow the development of the diffraction pattern, when tilting from an exact zone axis towards the 3-beam case with $\mathbf{G} = (200)$ in small steps of the tilt. The incoming beam is gradually tilted from an exact zone axis $[001]$ direction towards the 3-beam orientation via $[0, 1, 40] \simeq 1.43$ degrees, $[0, 1, 20] \simeq 2.86$ degrees, $[0, 1, 10] \simeq 5.71$ degrees and $[016] \simeq 9.46$ degrees tilt. At the endpoints we also did calculations with manual selection of beams. For the exact zone axis case we included kinematically allowed \mathbf{g} -vectors from the zero-order Laue zone (ZOLZ) with hkl indices less than 4 (in total 25 \mathbf{g} -vectors) and for the final 3-beam orientation we included beams of up to $\pm 5\mathbf{G}$ (in total 11 \mathbf{g} -vectors).

The $w_{\mathbf{g}}$ cut-off was the same as in previous sections, the P_{\min} was set to 10^{-4} , leading to well converged diffraction patterns and maps of the magnetic signal, as demonstrated in Sec. IV B. We chose to demonstrate the results at a sample thickness 50nm, because at this thickness the Kikuchi patterns appear sharp enough, yet not so sharp that we would observe aliasing artefacts due to discrete grid of pixels.

The results are summarized in Fig. 5. At the zone axis orientation we see a rich pattern, consisting of Bragg spots and multitude of Kikuchi bands and lines. The Kikuchi pattern is even better visible on the map of the magnetic signal. Number of thin lines with varying intensity and sign form a rich symmetric structure. Now let's compare them with a calculation with 25 \mathbf{g} -vectors from ZOLZ. The diffraction pattern seems to display the same structure, except for higher intensity. A cautious eye would spot some differences in relative intensity around $\{220\}$ spots. More differences can be seen in the maps of the magnetic signal. Particularly at larger scattering

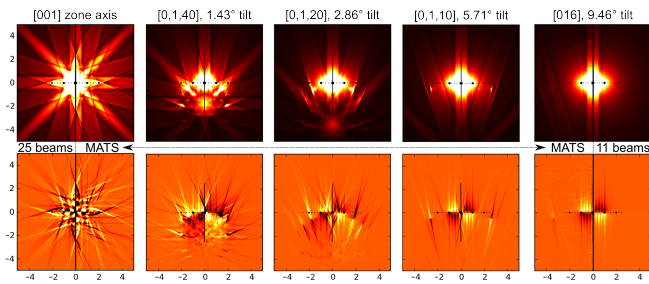


FIG. 5. Evolution of the diffraction pattern (top row) and of the map of magnetic signal (bottom row) when tilting the incoming beam from $[001]$ zone-axis orientation to 3-beam orientation $[016]$, which is approximately 9.5 degrees tilted from $[001]$. Intermediate orientations (beam directions $[0, 1, 40]$, $[0, 1, 20]$ and $[0, 1, 10]$) are indicated over the top row of panels. Left halfplane of $[001]$ zone axis orientation is calculated with 25-beams from the zero-order Laue zone, right half-plane of $[016]$ 3-beam orientation is calculated with 11 beams on the systematic row, and the other maps are calculated by MATS algorithm with $P_{\min} = 10^{-4}$. The q_x, q_y axes are in multiples of G , where $\mathbf{G} = (200)$. Calculations were performed on a mesh of 201×201 pixels.

angles, the pattern of Kikuchi lines is more rich in the MATS description.

As we tilt the beam, the pattern of beams is deforming, following the movement of the zone axis spot down. At beam tilt 1.43 degrees, corresponding to approximately 25 mrad, the zone axis spot moved down by a bit less than $2G$. The two-fold Bragg angle corresponding to $\mathbf{G} = (200)$ in bcc iron is 13.8 mrad, therefore the $2G \approx 27.6$ mrad, which agrees with the position of the zone axis spot in the map. The dominant Kikuchi bands are passing around this spot. In the map of the magnetic signal a noise-like signal forms around the zone axis center, and at the same time, there evolves a different dominant sign in the four quadrants of the diffraction plane. Both these trends continue with the tilt 2.86 degrees. The zone axis spot has weaker intensity, moves further down, to around $3.5G$ under the transmitted beam. At this angle and thickness, we see relatively strongly excited beams for $h = -3, -1, 1, 3$ and $k = -1$, i.e., spots under the systematic row of reflections. At the 5.7 degrees tilt the zone axis spot is outside the vertical range of our maps, we see a clean 3-beam pattern, with some intensities at $(\bar{1}\bar{1}0)$ and $(1\bar{1}0)$. Maps of the magnetic signal are considerably simpler, having a dominant sign in each quadrant. The four vertical lines represent Kikuchi lines. Skew lines at larger scattering angles show unexpectedly high magnetic signal, which might be a finding of potential practical importance.

At the final step, almost 10 degrees tilt the patterns complete the trends: three strongly excited spots along the systematic row, and three dominant Kikuchi bands. Magnetic signal showing the four vertical lines with signs corresponding to their quadrants. Interesting is a comparison to a systematic row approximation, where we

used only 11 \mathbf{g} -vectors – multiples of the $\mathbf{G} = (200)$. These patterns completely miss the skew Kikuchi bands and in the maps of the magnetic signal there is a very reduced pattern of lines.

D. Comparison between MATS and ICSC results

A similar but alternative computer program which calculates inner-shell ionization and backscattering cross sections for fast electrons incident on a crystal is available, known as the ICSC code, developed by Oxley and Allen³⁴. The program calculates the inelastic scattering coefficients for inner-shell ionization, pertinent to EELS and energy dispersive X-ray (EDX) analysis, using parameterizations of the atomic inelastic scattering factors. The program treats the dynamical scattering in a very similar way to MATS, but simplifies the calculations of dynamical form factors, using the approximation where the integration over all the final states of the scattered electron is replaced by an analytic expression, which unfortunately does not cover general MDFF calculation. As a consequence, the ICSC code only allows the EELS detector geometry concentric with the incident beam direction, i.e., the detector is aligned with the transmitted beam and the only variables are the sample orientation, convergence and collection angles. The ICSC code has been intensively tested for EDX results of atom location by channeling enhanced microanalysis (ALCHEMI), but little was published in testing the ICSC program for experimental EELS. It is thus a good chance for comparing the two methods.

As a benchmark we selected a cubic SiC crystal and the orientation dependent Si K-edge cross sections were compared. We have done two sets of calculations: 1) thickness dependence in a beam-rocking experiment in a systematic row orientation, and, 2) 2-dimensional beam-rocking in a $[001]$ s zone-axis orientation. In both cases, we selected parallel illumination and convergence angle of 10 mrad. The acceleration voltage was set to 300 keV. ICSC uses Doyle-Turner scattering factors³⁵, therefore for the sake of consistency, we used these also in calculations with the new BW code.

In the systematic row calculation, we assumed a beam tilt of approximately 10 degrees towards the $G_{(220)}$ systematic row conditions, i.e., the zone axis was $[\bar{1}18]$. For ICSC we used as set of 85 beams $(hkl) \perp [\bar{1}18]$. With the new BW code we did two sets of calculations - one with the fixed set of 9 beams from the systematic row (SR; up to $\pm 4G_{(220)}$) and another in the MATS approach with convergence parameter 10^{-4} . Results are summarized in Fig. 6. The results show good qualitative agreement, yet there are clear differences in details, even when comparing the ICSC calculation to a SR calculation. That indicates that approximations introduced in ICSC partially neglect some details of the dynamical diffraction process. Full MATS calculation shows expectedly even more of a fine structure, especially at larger thicknesses.

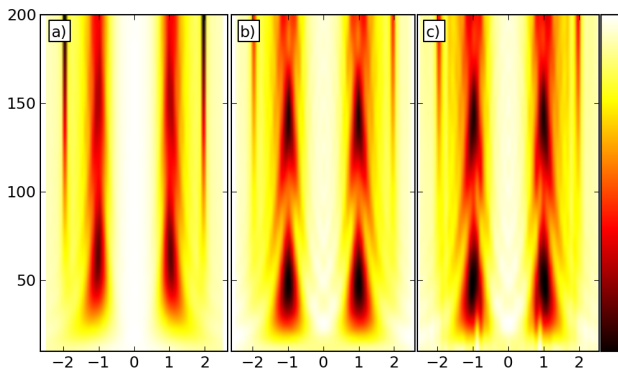


FIG. 6. Comparison of ICSC and MATS. Thickness profiles of K-edge of Si in SiC in systematic row orientation. Horizontal axis corresponds to a beam tilt in multiples of $\mathbf{G} = (220)$ and the vertical axis is thickness in nm. Calculations were performed for 51 tilts and 40 thicknesses. a) ICSC calculation, b) calculation with the new BW code, adopting the same parameters as ICSC, c) MATS calculation.

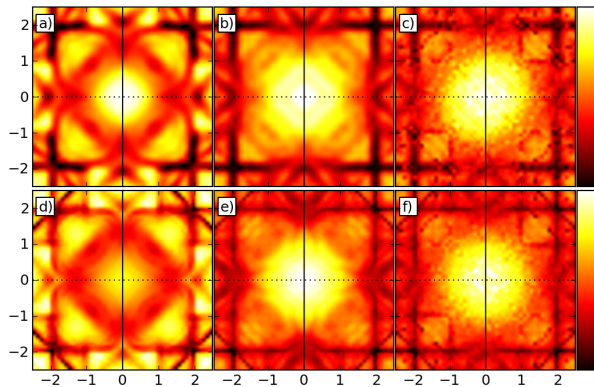


FIG. 7. Comparison of ICSC and MATS. Intensity of K-edge of Si in SiC in [001] zone-axis orientation as a function of beam rocking. Axes corresponds to a beam tilt in multiples of $\mathbf{G} = (200)$. Calculations were performed on a mesh of 51×51 pixels. a) ICSC calculation at 100 nm, b) calculation with the new BW code, adopting the same parameters as ICSC, c) MATS calculation at 100 nm, d)-f) the same as a)-c), but at 200 nm.

Detailed comparison to experiment would be needed to verify the fine features observed in MATS calculation.

The dependence of the double-differential scattering cross-section on beam rocking from the [001] zone axis orientation is shown in Fig. 7. Results by the ICSC code are shown in the left column. A set of 197 beams was used, all from the zero-order Laue zone. The same ones were used for comparison with the new BW code (shown in the middle column). Finally, results obtained by the

MATS method are shown in the right column. General qualitative features are in good agreement between all three computational approaches, nevertheless, as in the systematic-row orientation, also here we observe notable differences. Of particular interest is a soft breaking of the four-fold symmetry in the case of MATS calculations. It is easier to spot at 100 nm, where the central intensity maximum has a slightly prolate shape along the main diagonal of the pattern. This can be explained by presence of beams from higher-order Laue zones in the secular equation. They introduce (hkl) beams with $l \neq 0$ and due to the curvature of the Ewald sphere, (hkl) and $(hk\bar{l})$ have different excitation errors and thus their corresponding Bloch coefficients differ. In combination with the four-fold screw axis of this structure and a lack of inversion symmetry, the resulting beam-rocking pattern shows deviation from four-fold symmetry.

V. CONCLUSIONS

We have developed a new method for accurate summation over Bloch waves and their plane wave components (beams) named modified automatic term selection (MATS). The complexity of MATS scales inversely proportional to the cut-off for the term sizes. It allows highly accurate calculations at much lower computational costs compared to previous methods. We have demonstrated advantages of the method on capturing the intensity of the weakly excited spots outside the systematic row. The convergence properties of MATS were studied on a two-beam case simulation with focus on faint effects observed in EMCD experiments. A rich pattern of Kikuchi bands and lines was presented in a simulation of a tilting of the bcc crystal from the zone axis orientation to a three-beam orientation. We have also compared the new method to ICSC code and both display qualitatively the same results. In more detail, MATS calculation seems to provide more rich structures, most likely due to larger number of beams included in the secular equation.

VI. ACKNOWLEDGEMENTS

J.R. acknowledges the support of Swedish Research Council and STINT. Simulations were performed on computer cluster DORJE of Czech Academy of Sciences. This work was in part supported by the Grant-in-Aid for Scientific Research from MEXT, Japan. We thank the anonymous referees for their detailed comments and suggestions.

¹ W. H. Bragg, W. L. Bragg, Proc. Royal Soc. London A **88**, 428 (1913).

² M. von Laue, Annalen der Physik **42**, 971 (1913).

- ³ C. Davisson, L. H. Germer, *Phys. Rev.* **30**, 705 (1927).
- ⁴ G. P. Thomson, A. Reid, *Nature* **119**, 890 (1927).
- ⁵ S. Kikuchi, *Proc. Imp. Acad. Japan* **4**, 271, 275, 354, 471 (1928); *Jpn. J. Phys.* **5**, 83 (1928).
- ⁶ Y. Kainuma, *Acta Cryst. A* **8**, 24 (1955).
- ⁷ K. Okamoto, T. Ichinokawa, and Y.-H. Ohtsuki, *J. Phys. Soc. Jpn.* **30**, 1690 (1971).
- ⁸ R. F. Egerton, *Electron Energy Loss Spectroscopy in the Electron Microscope*, 3rd Edition, Springer, New York (2011), Chapter 4.1, p.231.
- ⁹ J. Rusz, S. Rubino, and P. Schattschneider, *Phys. Rev. B* **75**, 214425 (2007)
- ¹⁰ H. Kohl and H. Rose, *Advances in electronics and electron optics* **65**, 173 (1985).
- ¹¹ J. Rusz, S. Rubino, O. Eriksson, P. M. Oppeneer, K. Leifer, *Phys. Rev. B* **84**, 064444 (2011).
- ¹² M. Nelhiebel, P.-H. Louf, P. Schattschneider, P. Blaha, K. Schwarz, and B. Joffrey, *Phys. Rev. B* **59**, 12807 (1999).
- ¹³ L. Calmels, J. Rusz, *Ultramicroscopy* **110**, 1042 (2010).
- ¹⁴ K. Tatsumi, S. Muto, I. Nishida, J. Rusz, *Appl. Phys. Lett.* **96**, 201911 (2010).
- ¹⁵ M. Cowley and A. F. Moodie, *Acta Cryst.* **12**, 353-359 (1959).
- ¹⁶ K. Ishizuka and N. Uyeda, *Acta Cryst. A* **33**, 740-749 (1977).
- ¹⁷ M. Inokuti, *Rev. Mod. Phys.* **43**, 297-347 (1971).
- ¹⁸ A. J. F. Metherell, in *Electron Microscopy in Materials Science II*, 397 (1975), CEC, Luxembourg.
- ¹⁹ J. Verbeeck, C. Hébert, P. Schattschneider, S. Rubino, P. Novák, J. Rusz, F. Houdellier, and C. Gatel, *Ultramicroscopy* **108**, 865 (2008).
- ²⁰ P. Schattschneider, S. Rubino, M. Stöger-Pollach, C. Hébert, J. Rusz, and L. Calmels, *J. Appl. Phys.* **103**, 07D931 (2008).
- ²¹ L. J. Allen and T. W. Josselson, *Phys. Rev.* **52**, 3184-3198 (1995).
- ²² C. J. Rossouw, C. T. Forwood, M. A. Gibson, P. R. Miller, *Micron* **28**, 125137 (1997).
- ²³ Z. H. Levine, *Phys. Rev. B* **77**, 125314 (2008).
- ²⁴ K. Tatsumi and S. Muto, *J. Phys.: Condens. Matter.* **21**, 104213 (2009).
- ²⁵ E. J. Kirkland, *Advanced Computing in Electron Microscopy*, 2nd Ed., Springer, New York, Chapter 6.1, p.118.
- ²⁶ P. Schattschneider, M. Stöger-Pollach, S. Rubino, M. Sperl, Ch. Hurm, J. Zweck, and J. Rusz, *Phys. Rev. B* **78**, 104413 (2008).
- ²⁷ H. Lidbaum, J. Rusz, A. Liebig, B. Hjörvarsson, P. M. Oppeneer, E. Coronel, O. Eriksson, K. Leifer, *Phys. Rev. Lett.* **102**, 037201 (2009).
- ²⁸ P. Schattschneider, S. Rubino, C. Hébert, J. Rusz, J. Kuneš, P. Novák, E. Carlino, M. Fabrizioli, G. Panaccione and G. Rossi, *Nature* **441**, 486 (2006).
- ²⁹ J. Rusz, O. Eriksson, P. Novák, P. M. Oppeneer, *Phys. Rev. B* **76**, 060408(R) (2007).
- ³⁰ L. Calmels, F. Houdellier, B. Warot-Fonrose, C. Gatel, M. J. Hÿtch, V. Serin, E. Snoeck, P. Schattschneider, *Phys. Rev. B* **76**, 060409(R) (2007).
- ³¹ H. Lidbaum, J. Rusz, S. Rubino, A. Liebig, B. Hjörvarsson, P. M. Oppeneer, E. Coronel, O. Eriksson, and K. Leifer, *Ultramicroscopy* **110**, 1380 (2010).
- ³² J. Rusz, P. M. Oppeneer, H. Lidbaum, S. Rubino, K. Leifer, *J. Microscopy* **237**, 465 (2010).
- ³³ S. Rubino, P. Schattschneider, J. Rusz, J. Verbeeck, K. Leifer, *J. Phys. D: Appl. Phys.* **43**, 474005 (2010).
- ³⁴ M. P. Oxley and L. J. Allen, *J. Appl. Cryst.* **36**, 940-943 (2003); M. P. Oxley and L. J. Allen, *Acta Cryst. A* **56**, 470-490 (2000); M. P. Oxley and L. J. Allen, *Acta Cryst. A* **57**, 713-728 (2000).
- ³⁵ P. A. Doyle and P. S. Turner, *Acta Cryst. A* **24**, 390 (1968).