

1 **Long term efficacy and fate of a right ventricular outflow tract replacement using**
2 **a novel developed material with optimized biodegradation and elasticity**

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19 **ABSTRACT**

20 For decades, researchers have investigated the ideal material for clinical use in the

21 cardiovascular field. Several substitute materials are used clinically, but each has
22 drawbacks. Recently we developed poly(ϵ -caprolactone-co-D,L-lactide) (P(CL-DLLA))
23 polymers with optimized biodegradation and elasticity by adjusting the CL/DLLA
24 composition, and used these polymers in right ventricular outflow tract (RVOT)
25 replacement to evaluate long-term efficacy and outcomes. This P(CL-DLLA) material
26 was processed into a circular patch and used to replace a surgical defect in the RVOT of
27 adult rats. Control rats were implanted with expanded polytetrafluoroethylene (ePTFE).
28 Histologic evaluation was performed at 8, 24, and 48 weeks post-surgery. All animals
29 survived the surgery with no aneurysm formation or thrombus. In all periods, ePTFE
30 demonstrated fibrous tissue. In contrast, at 8 weeks P(CL-DLLA) showed infiltration of
31 macrophages and fibroblast-like cells into the remaining material. At 24 weeks,
32 P(CL-DLLA) was absorbed completely, and muscle-like tissue was present with
33 positive staining for α -sarcomeric actinin and cTnT. At 48 weeks, the cTnT-positive area
34 had increased. The P(CL-DLLA) with optimized elasticity and biodegradation induced
35 cardiac regeneration throughout the 48-week study period. Future application of this
36 material as a cardiovascular scaffold seems promising.

37

38 **Keywords:**

39 biodegradable polymer, elasticity, cardiac regeneration, myofibroblast, endothelization,
40 vascularization

41

42 **1. Introduction**

43 Some synthetic materials, such as polyethylene terephthalate fabric (DACRON) and
44 expanded polytetrafluoroethylene (ePTFE, e.g., Gore-Tex or IMPRA), are commonly
45 used for reconstruction of tissue deficiencies in cardiovascular surgery. These materials
46 are not clinically ideal since each has drawbacks, for instance unsuitable mechanical
47 properties for cardiac tissue, lack of biodegradation resulting in lack of native tissue
48 growth, and risks of calcification and infection.

49 Previously, we reported that applying a cardiac patch made of biodegradable
50 polyester urethane urea (PEUU) onto an infarcted area could prevent further cardiac
51 dilation and preserve cardiac function after myocardial infarction, due to the material's
52 suitable elasticity and strength [1-3]. Further, this material induced muscle
53 cellularization in which the cells had characteristics of early cardiomyocytes,
54 contributing to cardiac regeneration. Unfortunately, PEUU has not yet been approved by
55 the US Food and Drug Administration (FDA) for clinical usage.

56 We have focused on polylactic acid (PLA), poly- ϵ -caprolactone (PCL), and
57 polyglycolic acid (PGA), which have already been widely used clinically in the
58 construction of artificial bone, tendon, skin, and sutures, and are expected to be utilized
59 in the cardiovascular field. Our objective is to develop a biodegradable material with
60 suitable mechanical properties for cardiovascular reconstruction and to apply this

61 material to cardiac tissue in vivo to prove its long-term efficacy.

62 To create the cardiac biomaterial, we used four-armed
63 poly(ϵ -caprolactone-co-D,L-lactide) (i.e., P(CL-DLLA)), which was reported in
64 previous papers [4-6]. In general, PCL and PLA are quite mechanically rigid; however,
65 the elasticity of our P(CL-DLLA) material has been tuned for tissue compatibility and
66 biodegradability [6]. We were able to use this tunable material platform in novel
67 tissue-engineering scaffolds for nerve generation, spheroid culture, and the creation of
68 biomaterials for cancer therapy [7-9].

69 In this study, we optimized the material properties of P(CL-DLLA) for use as a
70 novel cardiac patch. The right ventricular outflow tract (RVOT) in rats was replaced
71 with P(CL-DLLA), and the RVOT material was then examined in terms of degradation,
72 angiogenesis, and endothelium and tissue formation after implantation periods of 8, 24,
73 and 48 weeks. It was thought that the endpoint of 48 weeks would demonstrate the fate
74 of the implanted biomaterial. This is the first study to report such long-term observation
75 of biodegradable material in the heart.

76

77 **2. Materials and methods**

78 *2.1. Experimental animals*

79 Adult male Sprague-Dawley rats (Japan SLC, Inc. Shizuoka, Japan) weighing 300 g
80 to 350 g were used for the RVOT replacement procedure. The research protocol

81 followed the National Institutes of Health guidelines for animal care and was approved
82 by the Institutional Animal Care and Use Committee of the Animal Experiment
83 Advisory Committee of the Nagoya University School of Medicine.

84

85 *2.2. Mechanical properties of P(CL-DLLA)*

86 Briefly, four-armed P(CL-DLLA) was synthesized by ring-opening polymerization
87 of CL and DLLA from terminal hydroxyl groups of pentaerythritol using tin octanoate
88 as a catalyst [7, 8]. Two types of the copolymer were also synthesized by adjusting the
89 CL/DLLA ratio to 80/20 and 60/40 (mol%). The ratio of CL/DLLA was confirmed by
90 ¹H NMR. The obtained copolymers were then reacted with acryloyl chloride to
91 introduce vinyl groups at the end chains. The end-functionalized macromonomers were
92 dissolved in xylene (50 wt%) and 2 wt% BPO was added. The macromonomer solution
93 was placed between two glass plates with a 4 cm x 4 cm Teflon spacer of 0.5 mm. The
94 glass plates were put in an oven at 80 °C for 3 h. The polymer was detached from the
95 glass plates and purified by immersion in a large amount of acetone. P(CL-DLLA) was
96 obtained after drying under reduced pressure. P(CL-DLLA) was characterized by tensile
97 testing (EZ-S500N, SHIMADZU, Kyoto, Japan) with a thermo-chamber that allowed
98 the temperature-dependent mechanical properties of samples to be determined. The
99 tensile tests were carried out at an elongation rate of 10 mm min⁻¹ at 25 °C and 37 °C,
100 and the elastic modulus was calculated from the initial slope of the stress–strain curve.

101 For the cyclic test, a single cyclic load was applied with a strain amplitude that was
102 gradually increased by 50% every cycle; the loading and unloading speeds were 10 mm
103 min⁻¹ at 25 °C until the test specimen failed. The morphology of the P(CL-DLLA)
104 which was thread a surgical suture was observed by scanning electron microscopy
105 (SEM; JCM-5000, JEOL).

106

107 *2.3. P(CL-DLLA) or ePTFE implantation for ROVT reconstruction*

108 The surgical procedure was based on the method previously reported by Sakai and
109 colleagues [10]. Briefly, the heart was exposed through median sternotomy and a
110 purse-string suture was placed in the RVOT free wall with 7-0 polypropylene to form a
111 perimeter with a diameter greater than 6 mm (Ethicon, Somerville, NJ). Suture ends
112 were passed through a 26-gauge plastic vascular cannula (TERUMO, Tokyo, Japan) and
113 a tourniquet was applied and tightened. The RVOT wall inside the purse-string stitching
114 was then distended and resected to create a defect just under 6 mm in diameter.
115 Six-mm-diameter P(CL-DLLA) or ePTFE (GORE-TEX cardiovascular patch, W. L.
116 Gore & Associates, Inc., NY, USA) was sutured along the margin of the purse-string
117 suture with an over-and-over method with 7-0 polypropylene to cover the defect. The
118 tourniquet was then released and the purse-string suture was removed, leaving a
119 whole-wall-thickness defect just under 6 mm in diameter covered by the materials. The
120 chest incision was closed in layers with running sutures of 4-0 Vicryl (Ethicon).

121 At each scheduled explant time point (8, 24, and 48 weeks), animals were
122 administered 500 units of heparin under anesthesia with 3% isoflurane (FUJIFILM
123 Wako Pure Chemical Co., Osaka, Japan) and were then euthanized (n = 9 per group) by
124 intravenous injection of an overdose of KCL solution. The heart was harvested and
125 frozen in OCT compound, which was pre-fixed in 4% paraformaldehyde for 48 h at
126 4 °C and gradually dehydrated in 10%, 15%, or 20% sucrose buffer overnight at 4 °C.

127

128 *2.4. Histology and immunohistochemistry*

129 The frozen heart tissue was serially cryosectioned into 10- μ m-thick specimens and
130 processed for Masson trichrome or immunohistochemical evaluation. Specimens for
131 immunohistochemistry were activated with HistoVT One (Nakalai Tesque, Kyoto,
132 Japan) antigen solution for 20 min at 70 °C. Slides were then reacted with antibodies
133 against α -sarcomeric actinin (mouse monoclonal 1:50, Abcam), α -smooth muscle actin
134 (α -SMA; rabbit polyclonal 1:500, mouse monoclonal 1:100, Abcam), von Willebrand
135 factor (vWF; rabbit polyclonal 1:100, Abcam), CD11b (rabbit polyclonal 1:200, Novus
136 Biologicals, Centennial, CO, USA), or cardiac troponin T (cTnT; rabbit polyclonal
137 1:400, Abcam) overnight at 4 °C. Secondary antibodies were Alexa Fluor
138 488-conjugated antibody (anti-mouse or anti-rabbit IgG (H+L), 1:5000, Cell Signaling,
139 Danvers, MA, USA) and Alexa Fluor 546-conjugated antibody (anti-rat or anti-rabbit
140 IgG (H+L), 1:5000, Cell Signaling). Nuclei were stained with 4',6-diamidino-2-

141 phenylindole, DAPI Fluoromount-G (SouthernBiotech, Birmingham, AL, USA).

142 Sections were also examined for the formation of a fibrous capsule around the
143 materials and observed with an FSX100 microscope (Olympus, Tokyo, Japan). For the
144 measurement of cTnT-positive areas, blood vessel density, and the number of
145 CD11b-positive cells at each replacement site, excluding autologous tissue, two
146 different sample sections were quantified by Image J software (NIH, Bethesda, MD,
147 USA). Vessels were identified as tubular structures that stained positively for vWF.

148

149 *2.5. Statistical analysis*

150 Statistical significance among the groups was determined by a one-way factorial
151 analysis of variance (ANOVA) using GraphPad Prism for Mac (Version 6, San Diego,
152 CA, USA). Experimental results are expressed as mean \pm S.D.

153

154 **3. Results**

155 *3.1. Characterization of P(CL-DLLA)*

156 Figure 1a shows the stress-strain curve of crosslinked P(CL-DLLA). The curve
157 with a CL/DLLA ratio of 80/20 was strongly temperature-dependent, and the elastic
158 moduli were 76.5 MPa at 25 °C and 307 kPa at 37 °C, respectively. On the other hand,
159 the curve with a CL/DLLA ratio of 60/40 showed polymeric, rubber-like stress-strain
160 curves, with elastic moduli of 241 kPa at 25 °C and 218 kPa at 37 °C, respectively.

161 Figure 1b shows a photograph of the crosslinked P(CL-DLLA) with a CL/DLLA ratio
162 of 60/40. The P(CL-DLLA) thickness estimated from the cross-sectional image was
163 approximately 0.38 mm, which was almost equivalent to that of ePTFE (0.4 mm) (Fig.
164 1c), and the P(CL-DLLA) had a smooth surface (Fig. 1d). A temperature-independent
165 mechanical property of P(CL-DLLA) with a CL/DLLA ratio of 60/40 was observed,
166 suggesting its amorphous nature. The cyclic tensile test, a single cyclic load was applied
167 with a strain amplitude that was gradually increased by 50% every cycle, showed that
168 The P(CL-DLLA) was failed at 1050% strain, indicating the material's super-elastic
169 nature (Fig. 1e). On the other hand, the result of the cyclic tensile test for P(CL-DLLA)
170 with a CL/DLLA ratio of 80/20 showed more a plastic-like nature, with an asymmetric
171 curve similar to that of ePTFE (data not shown).

172 Figure 2 shows that the P(CL-DLLA) with a CL/DLLA ratio of 60/40 returned to its
173 original form after a suture was passed through it (Fig. 2a). The morphologies of the
174 holes made in the P(CL-DLLA) and ePTFE were not significantly different, as shown in
175 Fig. 2b. On the other hand, when the suture was removed, the hole in the ePTFE
176 remained open while the P(CL-DLLA) returned to its original state (Fig. 2c).

177

178 3.2. *Intra- and postoperative courses*

179 In term of surgical handling, both the P(CL-DLLA) with a CL/DLLA ratio of 80/20
180 and the ePTFE were stiff and difficult to penetrate with the suture needle. On the other

181 hand, P(CL-DLLA) with a CL/DLLA ratio of 60/40 showed excellent surgical handling
182 and hemostasis, and no dehiscence during and after the continuous 7-0 prolene suture.
183 There were no postoperative deaths throughout the 48-week study period, and no
184 thrombosis occurred in either surgical group. In addition, neither group showed any
185 dehiscence or aneurysm formation at the implanted site. The heart surfaces in both
186 surgical groups were covered in connective tissue (Fig. 3).

187

188 *3.3. Histological observations*

189 Histological sections were stained with hematoxylin and eosin and Masson
190 trichrome. As shown in Fig. 4, the ePTFE (Fig. 4a-f) and P(CL-DLLA) (Fig. 4g, j)
191 groups at 8 weeks demonstrated inflammatory cells infiltration of the surrounding
192 layered fibrous tissue. No changes were observed in the ePTFE group throughout 48
193 weeks. In contrast, at 8 weeks the P(CL-DLLA) group showed remaining P(CL-DLLA)
194 as indicated by arrows, as well as infiltration of inflammatory cells and fibroblast-like
195 cells (Fig. 4g, j). At 24 weeks, the foreign body reaction had ended and the
196 P(CL-DLLA) was completely absorbed (Fig. 4h, k). In addition, muscle-like tissue and
197 collagen synthesis appeared at 24 and 48 weeks (Fig. 4i, l). Hematoxylin and eosin
198 staining indicated that there was no calcification in any of the samples.

199

200 *3.4. Expression patterns of α -sarcomeric actinin, α -SMA, cTnT, vWF, and CD11b in the*

201 *P(CL-DLLA) group*

202 In immunohistochemical analysis of α -sarcomeric actinin and α -SMA at 8 weeks,
203 the P(CL-DLLA) contained many cells with positive for α -SMA (Fig. 5a-c). At 24
204 weeks, immature α -sarcomeric actinin-positive cells appeared and tubular structures that
205 stained positively for α -SMA (Fig. 5d-f). At 48 weeks, abundant α -sarcomeric
206 actinin-positive cells were observed within the muscle bundles (Fig. 5g-i).

207 To confirm cTnT expression patterns, each section was co-stained with α -sarcomeric
208 actinin and the cardiac-specific protein cTnT. No tissues expressing both proteins were
209 seen at 8 weeks (Fig. 6a-c), while such tissues had begun to develop at 24 weeks (Fig.
210 6d-f) and were widespread at 48 weeks (Fig. 6g-i). The cTnT-positive area in the
211 replacement site increased gradually as time passed (Fig. 6j; 8 weeks, $5.6 \pm 3.3 \text{ mm}^2$; 24
212 weeks, $219.2 \pm 165.8 \text{ mm}^2$; 48 weeks, $416.6 \pm 275.5 \text{ mm}^2$). There was a significant
213 difference between each pair of time points (8 vs 24 weeks, $p < 0.01$; 8 vs 48 weeks, $p <$
214 0.001 ; 24 vs 48 weeks, $p < 0.01$).

215 For assessment of endothelialization and vascularization, samples were stained with
216 vWF as shown in Fig. 7. The P(CL-DLLA) group at each time point after implantation
217 showed complete endocardial endothelialization. Vascularization was observed at each
218 time point, and the maximum blood vessel density occurred at 8 weeks (Fig. 7j; 8 weeks,
219 $230.1 \pm 81.4 /\text{mm}^2$; 24 weeks, $119.8 \pm 39.9 /\text{mm}^2$; 48 weeks, $120.2 \pm 42.5 /\text{mm}^2$; 8 vs 24
220 weeks, $p < 0.001$; 8 vs 48 weeks, $p < 0.001$).

221 To identify infiltrating inflammatory cells, the sections were stained with CD11b.
222 Many CD11b-positive cells infiltrated the P(CL-DLLA), and their number decreased
223 over time (Fig. 8j; 8 weeks, 23.3 ± 7.9 %; 24 weeks, 8.0 ± 3.4 %; 48 weeks, $6.2 \pm$
224 1.7 %; 8 vs 24 weeks, $p < 0.001$; 8 vs 48 weeks, $p < 0.001$).

225

226 **4. Discussion**

227 We previously developed a novel PCL-based material with tunable thermal and
228 mechanical properties as well as shape memory ability [8, 11, 12]. Despite being
229 composed of only crosslinked P(CL-DLLA), this biodegradable polymeric material
230 possesses high elasticity and strength. We already utilized this material platform as a
231 scaffold for nerve regeneration and spheroid cell culture [7, 8]. Given this material's
232 superior characteristics, we assumed it could serve as an elastic patch for tissue
233 reconstruction in the cardiovascular system, but its in vivo biocompatibility,
234 biodegradability, and clinical potential remained unknown. Thus, we designed a new
235 P(CL-DLLA) material with different CL/DLLA composition for in vivo cardiac patch
236 application. The material's physical appearance, such as its surface morphology and
237 thickness, as well as its thermal and mechanical properties, were optimized by changing
238 the CL/DLLA ratio. A dramatic improvement in the material's characteristics, including
239 its mechanical properties and surgical handling, was achieved by small change (~ 20
240 mol%) in the CL/DLLA ratio. The optimized material, crosslinked P(CL-DLLA) with a

241 CL/DLLA ratio of 60/40, had tissue-compatible, super-elastic nature (< 300 kPa)
242 showing elastic behavior over 1000% strain, and we therefore anticipated that it would
243 be useful as a material for a cardiac patch.

244 The results of this study showed that when this novel, optimized P(CL-DLLA) was
245 used to repair a surgical defect in the RVOT, it achieved good initial hemostasis and
246 surgical handling, gradually degraded as cell migration and vascularization occurred,
247 and led to formation of completely new tissue by self-organization of the host cardiac
248 muscle-like cells. Importantly, this study specifically involved long-term observation
249 that has not been detailed in other reports.

250 When ePTFE was implanted in the RVOT area, it remained surrounded by loose
251 fibrous tissue for the entire 48-week study, and the repaired area was presumably too
252 stiff to serve as a functional heart wall. On the other hand, at 8 weeks the right ventricle
253 wall repaired with P(CL-DLLA) demonstrated marked cell migration and was found to
254 express α -SMA and CD11b. At all time points, the P(CL-DLLA) showed
255 endothelialization with vWF-positive cells as well as abundant vascularization with
256 co-expression of α -SMA and vWF, without any evidence of thrombosis. At 24 weeks,
257 the P(CL-DLLA) had completely degraded and was replaced by a muscle layer
258 consisting of cells that co-expressed α -sarcomeric actinin and cTnT, possibly
259 representing cardiomyocytes. Notably, at 48 weeks this muscle tissue was still present
260 and the number of cTnT-positive cells had increased. It is speculated that the substrate

261 stiffness and strain induced differentiation or de-differentiation (reverse-remodeling) of
262 the myofibroblast phenotype [13-15]. Further investigation is necessary to elucidate the
263 mechanism of this cardiac regeneration.

264 The elastic modulus values of native adult rat hearts have been reported to range
265 from 11.9 to 46.2 kPa (mean value of 25.6 kPa) [16], while that of our material in an in
266 vitro experiment was 218 kPa at 37 °C, which was sufficient for mechanical
267 reinforcement. According to Hazeltine et al [17], however, a 50-kPa elastic modulus is
268 preferred for cardiogenesis. It should be taken into consideration that in vivo, the
269 replacement material is degraded not only through hydrolysis but also through
270 macrophage phagocytosis. Significant tensile strength is required in the high-pressure
271 environment of the heart while the material is being replaced by new tissue. The
272 P(CL-DLLA) in this study functioned well clinically given the fact that no rats died
273 even over the long 48-week observation period. As shown in Fig. 4, the P(CL-DLLA) in
274 vivo was gradually fragmented, and this fragmentation might participate the change of
275 mechanical properties. In fact, our previous study indicated that the microchannel
276 structure of the P(CL-DLLA) scaffold affected its mechanical properties [7]. This report
277 shows that since the elastic modulus of P(CL-DLLA) gradually decreased as a result of
278 fragmentation in vivo, it might eventually be suitable to facilitate transformation into
279 differentiation of cardiomyocytes from myofibroblasts .

280 In this study, we optimized the unique characteristics of P(CL-DLLA), including its

281 elasticity and biodegradability, and thereby induced muscle tissue regeneration. At the
282 same time, several issues should be considered in further studies. First, the beneficial
283 effects of RVOT remodeling were observed in a rat model. Further exploration in a
284 large animal model is needed, where changes in the replaced RVOT area and the
285 volume of RVOT remodeling more closely approximate the clinical setting. Second, the
286 origin of the myofibroblasts that appeared in the replaced RVOT wall was unknown.
287 Histological sections did not clearly indicate that this tissue migrated from the healthy
288 periphery of the heart. Employment of a marrow ablation model using a chimera with
289 fluorescently labeled reconstituted marrow would permit investigation of these cells'
290 origin and differentiation. Finally, this cardiac regeneration using P(CL-DLLA) could be
291 more beneficial for left ventricular infarction. In our previous studies, cardiac function
292 after infarction could be preserved by covering the infarcted area with PEUU or by
293 injecting poly(NIPAAm-co-AAc-co-HEMAPTMC) hydrogel [1-3, 18]. Even though
294 further optimization is needed for a system with higher pressure than that of the right
295 ventricle, it is encouraging that "off-the-shelf" P(CL-DLLA) consisting of
296 FDA-approved material is quite useful in various clinical settings, especially for
297 preventing the need for biological treatment.

298

299 **Conclusion**

300 This study evaluated the potential of a novel cardiac patch made of biodegradable

301 cross-linked P(CL-DLLA) polymers. This patch had unique mechanical properties,
302 particularly optimized elasticity, that made it suitable for long-term, in vivo RVOT
303 repair. The P(CL-DLLA) successfully induced new tissue growth during the initial
304 24-week period, including differentiation of cardiomyocytes from myofibroblasts,
305 endocardial endothelialization, and vascularization, and the new tissue was maintained
306 throughout the 48-week study. P(CL-DLLA) warrants further investigation for the
307 reconstruction of tissue deficiencies in cardiovascular surgery.

308

309 **Declaration of competing interest**

310 The authors declare no conflicts of interest.

311

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323

324 **References**

325 [1] K.L. Fujimoto, K. Tobita, W.D. Merryman, J. Guan, N. Momoi, D.B. Stolz, M.S.
326 Sacks, B.B. Keller, W.R. Wagner, An elastic, biodegradable cardiac patch induces
327 contractile smooth muscle and improves cardiac remodeling and function in subacute
328 myocardial infarction, *J. Am. Coll. Cardiol.* 49 (2007) 2292-2300.
329 <https://doi.org/10.1016/j.jacc.2007.02.050>

330 [2] K.L. Fujimoto, J. Guan, H. Oshima, T. Sakai, W.R. Wagner, In vivo evaluation of a
331 porous, elastic, biodegradable patch for reconstructive cardiac procedures, *Ann. Thorac.*
332 *Surg.* 83 (2007) 648-654. <https://doi.org/10.1016/j.athoracsur.2006.06.085>

333 [3] K.L. Fujimoto, K. Tobita, J. Guan, R. Hashizume, K. Takanari, C.M. Alfieri, K.E.
334 Yutzey, W.R. Wagner, Placement of an elastic biodegradable cardiac patch on a subacute
335 infarcted heart leads to cellularization with early developmental cardiomyocyte
336 characteristics, *J. Card. Fail.* 18 (2012) 585-595.
337 <https://doi.org/10.1016/j.cardfail.2012.05.006>

338 [4] T. Aoyagi, F. Miyata, Y. Nagase, Preparation of Cross-Linked Aliphatic Polyester
339 and Application to Thermoresponsive Material, *J. Control. Release.* 32 (1994) 87-96.

340 [5] H. Miyasako, K. Yamamoto, A. Nakao, T. Aoyagi, Preparation of cross-linked

341 poly[(epsilon-caprolactone)-co-lactide] and biocompatibility studies for tissue
342 engineering materials, *Macromol. Biosci.* 7 (2007) 76-83.
343 <https://doi.org/10.1002/mabi.200600188>

344 [6] T. Muroya, K. Yamamoto, T. Aoyagi, Degradation of cross-linked aliphatic polyester
345 composed of poly(epsilon-caprolactone-co-D,L-lactide) depending on the thermal
346 properties, *Polym. Degrad. Stabil.* 94 (2009) 285-290.
347 <https://doi.org/10.1016/j.polymdegradstab.2008.12.014>

348 [7] K. Uto, T. Muroya, M. Okamoto, H. Tanaka, T. Murase, M. Ebara, T. Aoyagi,
349 Design of super-elastic biodegradable scaffolds with longitudinally oriented
350 microchannels and optimization of the channel size for Schwann cell migration, *Sci.*
351 *Technol. Adv. Mater.* 13 (2012) 064207. <https://doi.org/10.1088/1468-6996/13/6/064207>

352 [8] K. Uto, S.S. Mano, T. Aoyagi, M. Ebara, Substrate Fluidity Regulates Cell Adhesion
353 and Morphology on Poly(epsilon-caprolactone)-Based Materials, *Acs. Biomater. Sci.*
354 *Eng.* 2 (2016) 446-453. <https://doi.org/10.1021/acsbiomaterials.6b00058>

355 [9] S.S. Mano, K. Uto, T. Aoyagi, M. Ebara, Fluidity of biodegradable substrate
356 regulates carcinoma cell behavior: A novel approach to cancer therapy, *AIMS. Mater.*
357 *Sci.* 3 (2016) 66-82. <https://doi.org/10.3934/matasci.2016.1.66>

358 [10] T. Sakai, R.K. Li, R.D. Weisel, D.A. Mickle, E.T. Kim, Z.Q. Jia, T.M. Yau, The fate
359 of a tissue-engineered cardiac graft in the right ventricular outflow tract of the rat, *J.*
360 *Thorac. Cardiovasc. Surg.* 121 (2001) 932-942.

361 <https://doi.org/10.1067/mtc.2001.113600>

362 [11] K. Uto, T. Aoyagi, C.A. DeForest, A.S. Hoffman, M. Ebara, A Combinational
363 Effect of "Bulk" and "Surface" Shape-Memory Transitions on the Regulation of Cell
364 Alignment, *Adv. Healthc. Mater.* 6 (2017). <https://doi.org/10.1002/adhm.201601439>

365 [12] K. Uto, M. Ebara, T. Aoyagi, Temperature-responsive poly(epsilon-caprolactone)
366 cell culture platform with dynamically tunable nano-roughness and elasticity for control
367 of myoblast morphology, *Int. J. Mol. Sci.* 15 (2014) 1511-1524.
368 <https://doi.org/10.3390/ijms15011511>

369 [13] A.K. Schroer, W.D. Merryman, Mechanobiology of myofibroblast adhesion in
370 fibrotic cardiac disease, *J. Cell. Sci.* 128 (2015) 1865-1875.
371 <https://doi.org/10.1242/jcs.162891>

372 [14] P. Kim, N. Chu, J. Davis, D.H. Kim, Mechanoregulation of Myofibroblast Fate and
373 Cardiac Fibrosis, *Adv. Biosyst.* 2 (2018). <https://doi.org/10.1002/adbi.201700172>

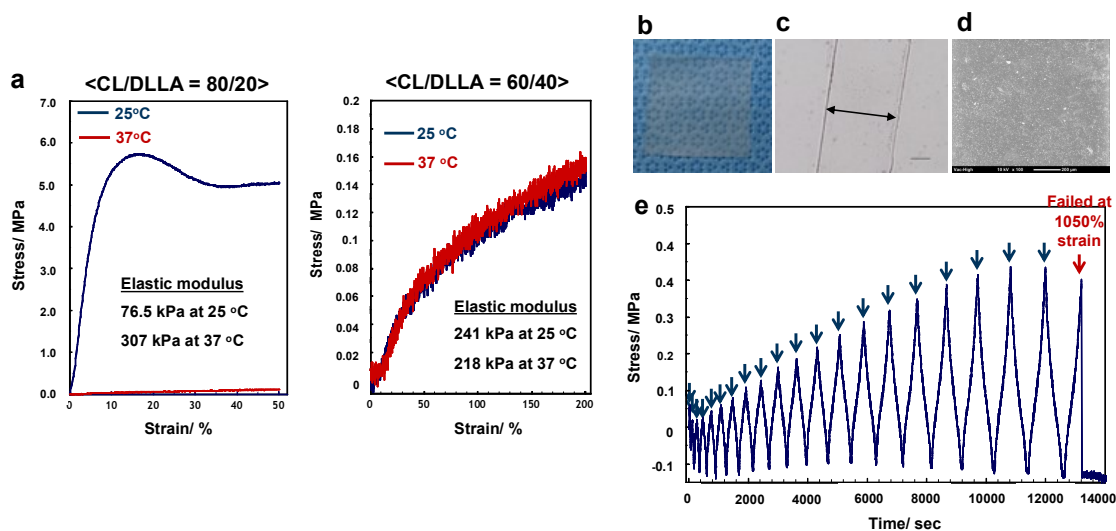
374 [15] C. Godbout, L. Follonier Castella, E.A. Smith, N. Talele, M.L. Chow, A. Garonna,
375 B. Hinz, The mechanical environment modulates intracellular calcium oscillation
376 activities of myofibroblasts, *PLoS One.* 8 (2013) e64560.
377 <https://doi.org/10.1371/journal.pone.0064560>

378 [16] B. Bhana, R.K. Iyer, W.L. Chen, R. Zhao, K.L. Sider, M. Likhitpanichkul, C.A.
379 Simmons, M. Radisic, Influence of substrate stiffness on the phenotype of heart cells,
380 *Biotechnol. Bioeng.* 105 (2010) 1148-1160. <https://doi.org/10.1002/bit.22647>

381 [17] L.B. Hazeltine, M.G. Badur, X.Lian, A Das, W. H, S.P. Palecek, Temporal Impact
 382 of Substrate Mechanics on Differentiation of Human Embryonic Stem Cells to
 383 Cardiomyocytes, Acta Biomater. 10 (2014) 604-612.
 384 <https://doi.org/10.1016/j.actbio.2013.10.033>
 385 [18] K.L. Fujimoto, Z. Ma, D.M. Nelson, R. Hashizume, J. Guan, K. Tobita, W.R.
 386 Wagner, Synthesis, characterization and therapeutic efficacy of a biodegradable,
 387 thermoresponsive hydrogel designed for application in chronic infarcted myocardium,
 388 Biomaterials. 30 (2009) 4357-4368. <https://doi.org/10.1016/j.biomaterials.2009.04.055>

389

390 **Figures**

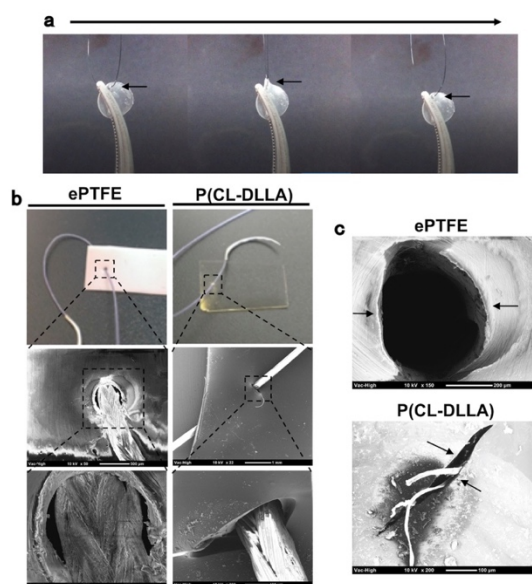


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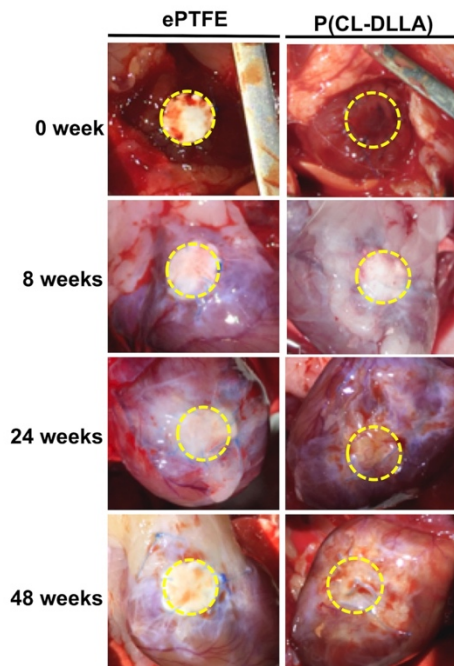
392

393 **Fig. 1.** Form and characterization of P(CL-DLLA). (a) Stress-strain curves of
 394 P(CL-DLLA) at 25 and 45 °C. (b) Photograph of P(CL-DLLA) with a CL/DLLA ratio

395 of 60/40 prepared by thermal crosslinking. (c) Cross-sectional image of P(CL-DLLA)
396 with a CL/DLLA ratio of 60/40 observed by an optical microscope (scale bar, 100 μm).
397 (d) Scanning electron microscopy (SEM) image of surface morphology of P(CL-DLLA)
398 with a CL/DLLA ratio of 60/40 (scale bar, 200 μm). (e) Cyclic tensile test with a
399 loading/unloading speed of 10 mm min⁻¹ at 25 °C. The constant strain amplitude of
400 50% per cycle was applied to the P(CL-DLLA) with a CL/DLLA ratio of 60/40 until it
401 broke.



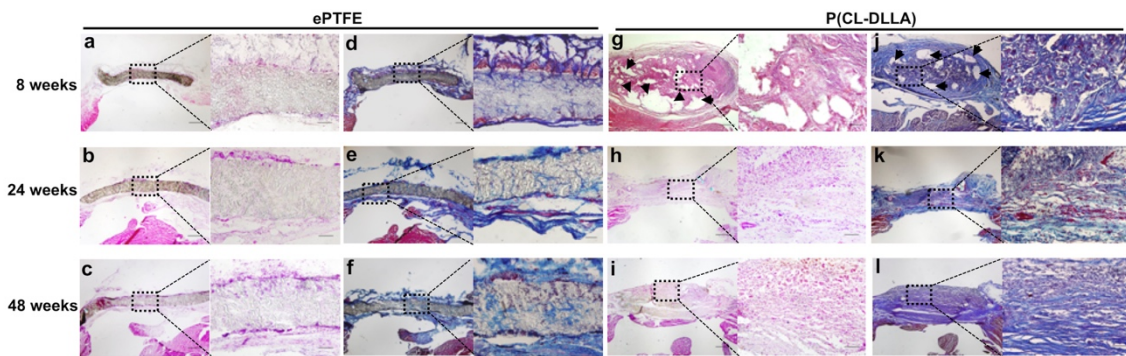
402
403 **Fig. 2.** Morphology of P(CL-DLLA) with a CL/DLLA ratio of 60/40. (a) Images over
404 time before and after pulling the surgical suture threaded on the P(CL-DLLA). (b)
405 Photographs and SEM images of the P(CL-DLLA) and ePTFE at the location of suture
406 penetration. (c) SEM images of P(CL-DLLA) and ePTFE surfaces after suture removal.



407

408 **Fig. 3.** RVOT repair with ePTFE or P(CL-DLLA). (a) Representative images at 0, 8, 24

409 and 48 weeks after RVOT repair with 6-mm-diameter P(CL-DLLA) or ePTFE.



410

411 **Fig. 4.** Time course of the healing response after RVOT repair with ePTFE or

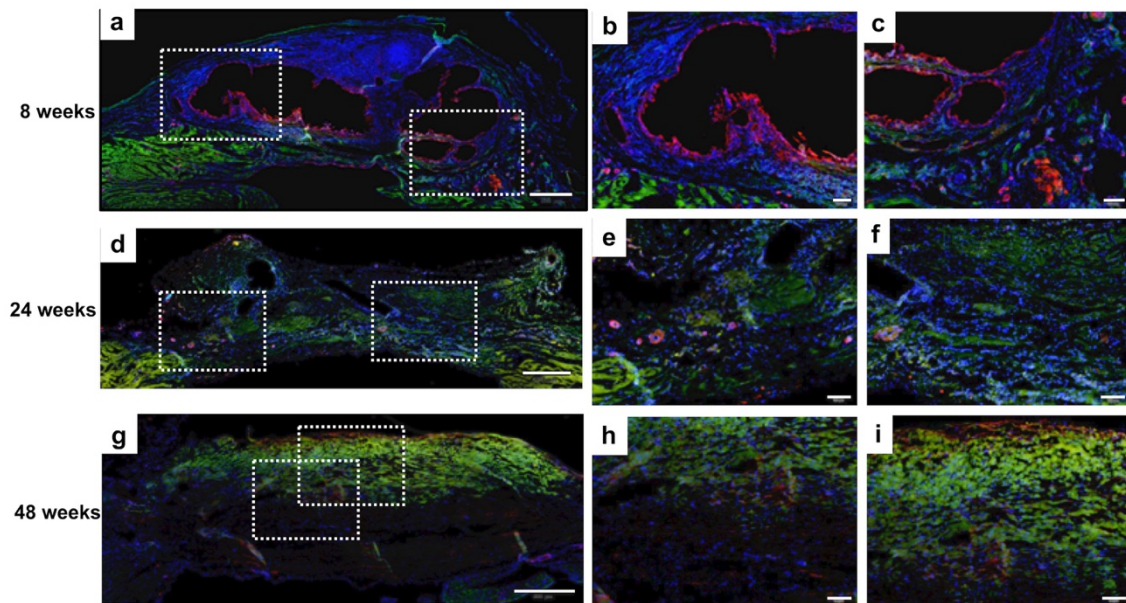
412 P(CL-DLLA). Representative images of the ePTFE group at 8 weeks (a, d), 24 weeks (b,

413 e), and 48 weeks (c, f), and the P(CL-DLLA) group at 8 weeks (g, j), 24 weeks (h, k),

414 and 48 weeks (i, l). Samples were stained with hematoxylin and eosin staining (a-f) or

415 Masson trichrome staining (g-l). Scale bars: 200 μm in the left images, and 50 μm in the

416 right images. Arrows indicate the area of remaining P(CL-DLLA).



417

418 **Fig. 5.** Time course of immunofluorescence staining of the P(CL-DLLA) area with

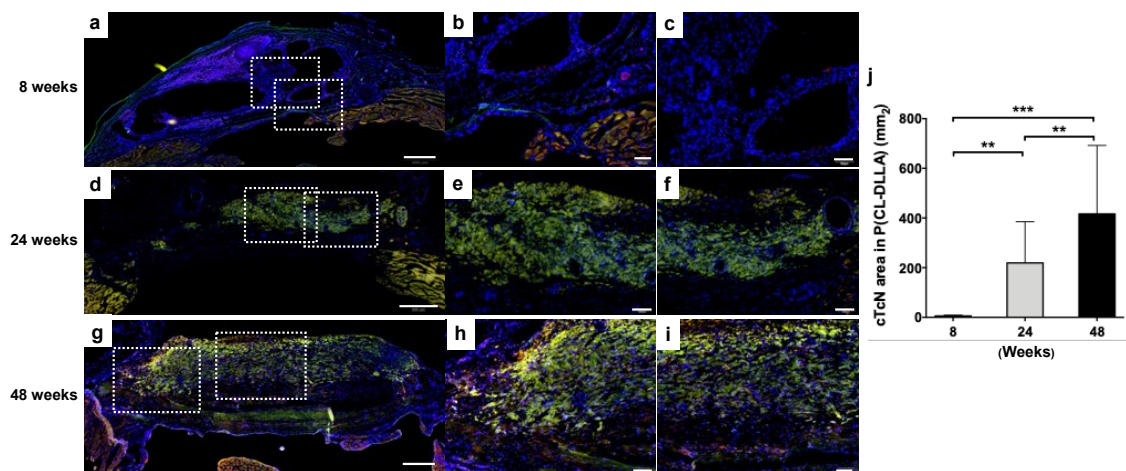
419 α -sarcomeric actinin and α -SMA. Representative images of the P(CL-DLLA) group at 8

420 weeks (a-c), 24 weeks (d-f), and 48 weeks (g-i). α -sarcomeric actinin staining appears

421 green, α -SMA staining appears red, and nuclear staining appears blue. White squares

422 indicate areas with greater magnification in the panels to the right. Scale bars: 500 μ m

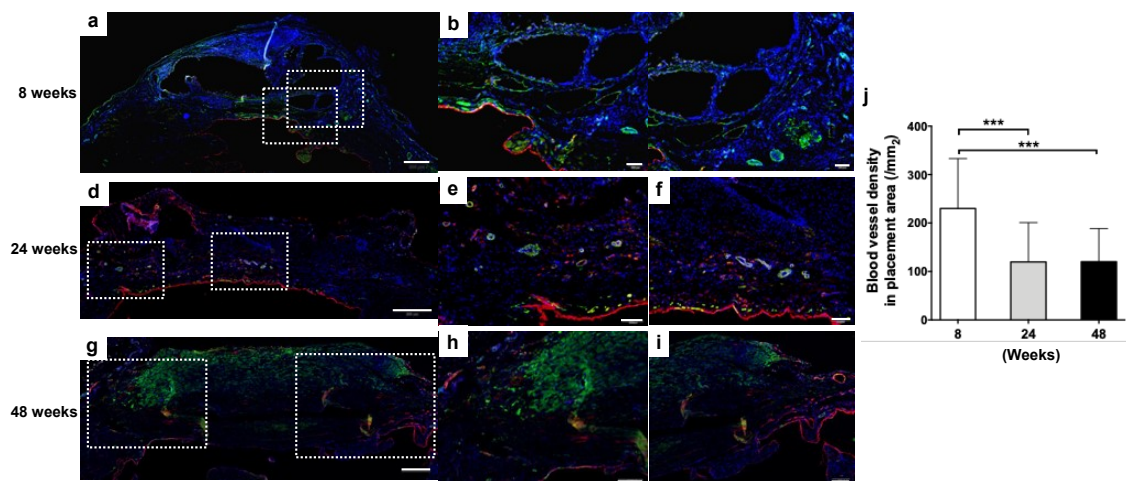
423 in (a, d, g), 100 μ m in (b, c, e, f, h, i).



424

Figure 6

425 **Fig. 6.** Time course of immunofluorescence staining of the P(CL-DLLA) area with
 426 α -sarcomeric actinin and cTnT. Representative images of the P(CL-DLLA) group at 8
 427 weeks (a-c), 24 weeks (d-f), and 48 weeks (g-i). Alpha-sarcomeric actinin staining
 428 appears green, cTnT staining appears red, and nuclear staining appears blue. White
 429 squares indicate areas with greater magnification in the panels to the right. Scale bars:
 430 500 μ m in (a, d, g), 100 μ m in (b, c, e, f, h, i). (j) The cTnT-positive area in the
 431 P(CL-DLLA) repair at 8, 24, and 48 weeks. Data are means \pm S.D. ** $p < 0.01$ and *** p
 432 < 0.001 assessed by one-way ANOVA.

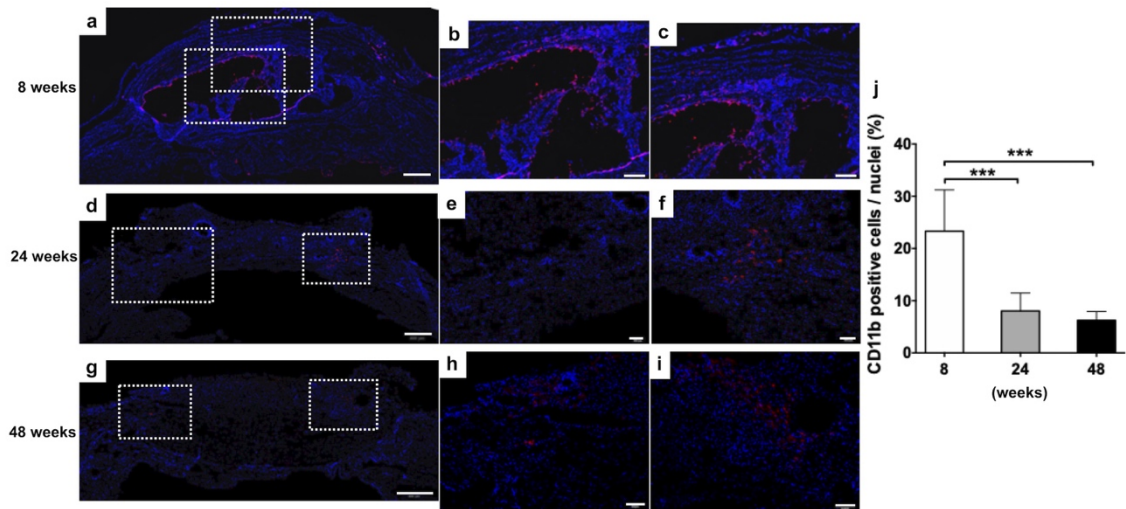


433 **Figure 7**

434 **Fig. 7.** Time course of immunofluorescence staining of the P(CL-DLLA) area with
 435 α -SMA and vWF. Representative images of the P(CL-DLLA) group at 8 weeks (a-c), 24
 436 weeks (d-f), and 48 weeks (g-i). Alpha-SMA staining appears green, vWF staining
 437 appears red, and nuclear staining appears blue. White squares indicate areas with greater
 438 magnification in the panels to the right. Scale bars: 500 μ m in (a, d, g), 100 μ m in (b, c,

439 e, f, h, i). (j) The blood vessel density in the P(CL-DLLA) repair at 8, 24, and 48 weeks.

440 Data are means \pm S.D. *** $p < 0.001$ assessed by one-way ANOVA.



441

442 **Fig. 8.** Time course of immunofluorescence staining of the P(CL-DLLA) area with

443 CD11b. Representative images of the P(CL-DLLA) group at 8 weeks (a-c), 24 weeks

444 (d-f), and 48 weeks (g-i). CD11b staining appears red and nuclear staining appears blue.

445 White squares indicate areas with greater magnification in the panels to the right. Scale

446 bars: 500 μm in (a, d, g), 100 μm in (b, c, e, f, h, i). (j) The number of CD11b-positive

447 cells in the P(CL-DLLA) repair at 8, 24, and 48 weeks. Data are means \pm S.D. *** $p <$

448 0.001 assessed by one-way ANOVA.