# Electric-field modulation of perpendicular magnetic anisotropy and damping constant in MgO/Co/Pt trilayers

A. Sakoguchi(砂古口藍子)<sup>1</sup>, D. Oshima(大島大輝)<sup>2</sup>, S. Iwata(岩田聡)<sup>3</sup> and T. Kato (加藤剛志)<sup>2</sup>

<sup>1</sup>Department of Electronics, Nagoya University, Nagoya 464-8603, Japan

<sup>2</sup>Institute of Materials and Systems for Sustainability, Nagoya University, Nagoya, 464-8603, Japan <sup>3</sup>Nagoya Industrial Science Research Institute, Nagoya 464-0819, Japan

Electric field modulations of the perpendicular magnetic anisotropy (PMA) and magnetization dynamics of MgO/Co/Pt trilayers were measured by time-resolved magneto-optical Kerr effect (TRMOKE). The anisotropy field estimated from the anomalous Hall effect applying a field along in-plane was confirmed to be modified by the electric field, which results in the PMA variation with electric field of -32 fJ/(V·m). The PMA variation was also confirmed from TRMOKE measurements, which was estimated to be -47fJ/(V·m). From TRMOKE measurements, we also observed the reduction of the inverse of relaxation time  $1/\tau$  of the magnetization precession with increasing the electric field. The  $1/\tau$  was analyzed by assuming the contributions of effective Gilbert damping  $\alpha_{eff}$ , anisotropy distribution, anisotropy angle distribution, and two magnon scattering, and we found  $\alpha_{eff}$  decreased with increasing the electric field whose modulation was -0.0046 nm/V, corresponding to -13% per 1 V/nm. These results are similar to the electric field effect in (001)-orientated MgO/CoFeB/Ta trilayers (A. Okada *et al.*, Appl. Phys. Lett. 105, 052415 (2014)), however our finding indicates similar PMA and  $\alpha$  modulations are found in MgO/Co/Pt trilayers having MgO(111)/fcc-Co interface.

Index Terms-Gilbert damping, Magnetic anisotropy, Electric field, MgO/Co/Pt

## I. INTRODUCTION

JOLTAGE-TORQUE switching has been attracted much attention as a writing method in magnetic random access memories (MRAM) with low power consumption. The writing energy in the voltage-torque switching is estimated to be 1 fJ/bit, which is about 1/500 of the switching energy for the conventional spin transfer torque (STT) switching. In the voltage-torque switching, the magnetization precession is triggered by the change of perpendicular magnetic anisotropy (PMA) by applying a voltage pulse, and the magnetization switching completes by half period of the magnetization precession [1]. Matsumoto et al. pointed out tunings of PMA and Gilbert damping are crucial to reduce the writing error in voltage-torque MRAM [2]. Thus, for the stable magnetization switching, the electric field dependence of PMA as well as magnetization dynamics are both important topics to be understood. Many studies on the voltage-induced modulation of PMA has been carried out [3]-[5], whereas there are only a few reports on the magnetization dynamics under the application of voltage [6]-[8]. Okada et al. reported Gilbert damping of MgO/CoFeB/Ta system is modulated by the application of gate voltage [6], while Sasaki et al. did not observe the significant change of Gilbert damping of MgO/CoFeB/Ta system with the applied electric field [7].

In this study, we measured the magnetization dynamics by using time-resolved magneto-optical Kerr effect (TRMOKE) under the application of the electric field in MgO/Co/Pt trilayers which have a high PMA, and estimated the electric

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field modulation of the PMA and Gilbert damping. Large electric field modulation of the magnetic properties is reported in the MgO/Co/Pt trilayers [9], [10] due to the variation of electron numbers in the thin Co layer. The Gilbert damping is known to be sensitive to the density of states at the Fermi energy [11], [12], and thus large electric field modulation of the Gilbert damping due to the variation of electron number is expected in the MgO/Co/Pt system. Gilmore *et al.* reported the Gilbert damping is dependent on the crystallographic orientation due to the anisotropic Fermi surface [12]. The MgO/Co/Pt trilayer will have MgO(111)/fcc-Co interface which is different from MgO(001)/bcc-CoFe interface in the previously reported MgO/CoFeB/Ta trilayers. Therefore, it is interesting to compare the electric field modulation of PMA and damping of the MgO/Co/Pt trilayer with those of the previously reported MgO/CoFeB/Ta trilayers.

## II. EXPERIMENTAL METHOD

## $ITO(20)/MgO(10)/Co(t_{Co})/Pt(1.6)/Ta(10)/thermally$

oxidized Si substrate (thickness in nanometer) was prepared by an rf magnetron systems. Structure and hysteresis loops of the MgO/Co/Pt trilayers were characterized by X-ray diffraction and alternating gradient field magnetometer, respectively. To measure anomalous Hall effect under the application of electric field, the film was microfabricated in a Hall-bar structure and the 100  $\mu$ m squared pillar was fabricated at the center of the Hall-bar, and the gate electrode of ITO (15 nm) was deposited on the pillar (see Fig. 1). The gate voltage was applied from –3 V to 3 V corresponding to the electric field from –0.3 V/nm to 0.3 V/nm. Hysteresis loop along film normal direction was obtained from anomalous Hall effect (AHE). AHE was also measured under in-plane magnetic field to estimate the anisotropy field. The in-plane

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component of the magnetization  $M_{\rm inp}$  was calculated by using the relation;  $M_{\rm inp}/M_{\rm s} = \sin\{\cos^{-1}(R_{\rm H}/R_{\rm Hmax})\}$ , where  $M_{\rm s}$  is the saturation magnetization, and  $R_{\rm H}$  and  $R_{\rm Hmax}$  are Hall resistance and the maximum Hall resistance, respectively.

TRMOKE was measured by pump-probe method using high power fiber laser with a wavelength of  $\lambda = 1040$  nm, a pulse width of 500 fs and a repetition frequency of 100 kHz [13]-[15]. Pump beam with  $\lambda = 1040$  nm was incident to the film surface to excite the magnetization precession, and the magnetization precession was monitored by Kerr rotation angle of reflected probe beam with  $\lambda = 520$  nm. The external magnetic field  $H_{\text{ext}}$  in the range of 5-14 kOe was applied in the direction of 73 ° from the film normal. The angular frequency  $\omega$  and relaxation time  $\tau$  of the magnetization precession were extracted from the damped oscillation signals in TRMOKE. The  $H_{\text{ext}}$  dependence of  $\omega$  and  $\tau$  of the precession were fitted to estimate effective anisotropy field  $H_{\text{k}}^{\text{eff}}$ , g-factor and effective damping constant  $\alpha_{\text{eff}}$ .

FIG. 1 HERE (Note white space above and below.)

#### **III.** RESULT AND DISCUSSIONS

## A. Structural analysis

Figure 2 shows XRD profile of MgO(10)/Co(1.6)/Pt(1.6) trilayer. Beside the peak from Si substrate (shown as \* in the figure), ITO 222 and  $\beta$ -Ta 002 peaks are visible, and broad peak around  $2\theta = 40^{\circ}$  is due to the overlapping of fcc-Pt 111 and MgO 111. No peaks are seen near MgO 200 ( $2\theta = 43^{\circ}$ ) and Pt 200 ( $2\theta = 46^{\circ}$ ), and thus the present MgO/Co/Pt trilayer is considered to have MgO(111)/fcc-Co interface.

FIG. 2 HERE (Note white space above and below.)

## B. Anomalous Hall effect

The MgO(10)/Co(1.2)/Pt(1.6) trilayer exhibited large PMA, and the coercivity in perpendicular direction was 160 Oe, which were obtained from AHE loop applying a magnetic field along perpendicular to the film plane (not shown here). Figure 3 (a) shows the AHE of MgO(10)/Co(1.2)/Pt(1.6) trilayer applying in-plane magnetic field under the application of the gate voltage of  $\pm 1.5$  V. The in-plane component of the magnetization  $M_{inp}$  was calculated from Fig. 3 (a), and the linear behavior of in-plane *M*-*H* loops was observed as shown in Fig. 3 (b). The slope of in-plane *M*-*H* loops increased by applying a positive electric field. By the linear fitting of  $M_{inp}$ , effective anisotropy field  $H_k^{eff}$  was estimated to be  $H_k^{eff} =$ 4220 Oe for the electric field of E = 0.15 V/nm and  $H_k^{eff} =$ 4350 Oe for E = -0.15 V/nm. The electric field dependence of  $H_k^{eff}$  is discussed later in Fig. 4.

FIG. 3 HERE (Note white space above and below.)

## C. TRMOKE

Figure 4 (a) shows the typical magnetization dynamics of

MgO(10)/Co(1.2)/Pt(1.6) trilayer under the application of the gate voltage of  $\pm 3.0$  V. In Fig. 4 (a) exponentially decayed background in the raw TRMOKE data was subtracted [13]. The external magnetic field of <u>Hext</u> = 14 kOe was applied during the measurements. From Fig. 4 (a), both angular frequency  $\omega$  and the relaxation time  $\tau$  of the magnetization precession were confirmed to vary with the electric field. The effective anisotropy field  $H_k^{\text{eff}}$  was estimated from the following analytical expressions derived from Landau-Lifshitz-Gilbert equation [16], [17].

$$\omega = \gamma \sqrt{H_{X0} H_{Y0}},$$
  

$$H_{X0} = H_{\text{ext}} \cos(\theta_{\text{H}} - \theta) + H_{\text{k}}^{\text{eff}} \cos^2 \theta,$$
  

$$H_{Y0} = H_{\text{ext}} \cos(\theta_{\text{H}} - \theta) + H_{\text{k}}^{\text{eff}} \cos 2\theta,$$
  
(1)

where  $\gamma = \mu_{\rm B}g/\hbar = 1.105 \times 10^5 g \text{ (m/A·s)}$  is the gyromagnetic constant,  $\theta_{\rm H}$  is the direction of <u>*H*</u><sub>ext</sub> from the film normal, and  $\theta$  is the stable magnetization angle from the film normal calculated by minimizing the following magnetic energy,

$$E = -M_{\rm s}H_{\rm ext}\cos(\theta_{\rm H} - \theta) + \frac{M_{\rm s}H_{\rm k}^{\rm eff}}{2}\sin^2\theta.$$
 (2)

Figure 4 (b) is the electric field dependence of  $H_k^{\text{eff}}$  of the MgO(10)/Co(1.2)/Pt(1.6) trilayer estimated from TRMOKE.  $H_k^{\text{eff}}$  estimated from AHE was also shown as a comparison. There are some difference (roughly 10%) in the absolute value of  $H_k^{\text{eff}}$  between TRMOKE and AHE. This difference is considered to be due to the limited maximum field applied in AHE experiments (see Fig. 3). The electric field dependence of  $H_k^{\text{eff}}$  is similar in TRMOKE and AHE;  $H_k^{\text{eff}}$  decreased with increasing the electric field. The variation of PMA with electric field estimated from TRMOKE and AHE was -47 fJ/(V·m) and -32 fJ/(V·m), respectively.

## FIG. 4 HERE (Note white space above and below.)

Figure 5 (a) shows the electric field dependence of the inverse of the relaxation time  $1/\tau$  at  $H_{\text{ext}} = 14$  kOe estimated from TRMOKE. Just as  $H_k^{\text{eff}}$ ,  $1/\tau$  also exhibited significant electric field dependence. Figure 5 (b) shows the magnetic field dependence of  $1/\tau$  of MgO(10)/Co(1.2)/Pt(1.6) trilayer under E = 0.3 V/nm. Since  $1/\tau$  is known to include Gilbert damping as well as the extrinsic contributions, such as anisotropy distribution and two magnon scattering (TMS) [17], the external field dependence of  $1/\tau$  shown as closed circles in Fig. 5 (b) was fitted by assuming four contributions; effective Gilbert damping  $\alpha_{\text{eff}}$ , anisotropy axis distribution  $\Delta\theta_{\text{H}}$ , anisotropy distribution  $\Delta H_k^{\text{eff}}$  and two magnon scattering (TMS) [17]-[21].

$$\frac{1}{\tau} = \frac{1}{\tau_{\text{Gilbert}}} + \frac{1}{\tau_{\Delta\theta_{\text{H}}}} + \frac{1}{\tau_{\Delta H_{\text{k}}^{\text{eff}}}} + \frac{1}{\tau_{\text{TMS}}} \\ = \alpha_{\text{eff}}\omega + \frac{1}{2} \left| \frac{\partial\omega}{\partial\theta_{\text{H}}} \right| \Delta\theta_{\text{H}} + \frac{1}{2} \left| \frac{\partial\omega}{\partial H_{\text{k}}^{\text{eff}}} \right| \Delta H_{\text{k}}^{\text{eff}}$$

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$$+N_0 \sum_{\mathbf{k}} \frac{C(k)}{\omega} \operatorname{Im}\left(\frac{1}{\omega_{\mathbf{k}}^2 - \omega^2 + i\omega\delta\omega_{\mathbf{k}}}\right),\tag{3}$$

where  $N_0$ , C(k),  $\omega_k$  and  $\delta\omega_k$  are the intensity of two magnon scattering, the correlation function, spin wave dispersion and inverse life time of the spin wave, respectively. Here, it was assumed that only  $H_k^{\text{eff}}$  and  $\alpha_{\text{eff}}$  depend on the electric field to reproduce the experimental results.  $\Delta H_k^{\text{eff}}$  and  $1/\tau_{\text{TMS}}$  depend on  $H_k^{\text{eff}}$  and thus they depend on the electric field through the variation of  $H_k^{\text{eff}}$ . Details of the fitting are described in Appendix.

## FIG. 5 HERE (Note white space above and below.)

Figure 6 (a) shows the electric field dependence of four constituents in (3) at  $H_{\text{ext}} = 14$  kOe. Two contributions, effective damping  $\alpha_{\rm eff}$  and the extrinsic term of  $\Delta \theta_{\rm H}$ , were found to be dominant on the relaxation of the precession, and these terms decreased with increasing the electric field. Thus, it is concluded that the electric field dependence of  $1/\tau$  results from the sum of the electric field modulation of these two contributions. Figure 6 (b) shows the electric field dependence of  $\alpha_{\rm eff}$ , and the  $\alpha_{\rm eff}$  value decreased with increasing electric field with a ratio of -0.0046 nm/V, corresponding to -13 % per 1 V/nm. Okada *et al.*, reported that damping constant  $\alpha$  of MgO/CoFeB/Ta trilayers decreased linearly with increasing electric field [6]. The ratio of  $\alpha$  modulation with the electric field was -0.003 nm/V, corresponding to -21 % per 1 V/nm. They also reported the decrease of PMA with increasing electric field with the variation of  $-60 \text{ fJ/(V} \cdot \text{m})$ . These results are similar to the results obtained in this study. Electric field modulation of the magnetic properties of metallic thin films is considered to originate from the variation of the electronic structure with a thickness of  $\sim 0.1$  nm from the interface due to short Thomas-Fermi screening length in metal. MgO/Co/Pt trilayer in this study has MgO(111)/fcc-Co interface as shown in Fig. 1, and we expected different electric field modulation of  $\alpha$  from that of previously reported MgO/CoFeB/Ta trilayers with MgO(001)/bcc-CoFe interface. In fact, the sign of the modulation of PMA was reported to depend on the underlayer [22], [23]. However, our results show that similar electric field dependence of PMA and  $\alpha$  between (111)-oriented MgO/Co/Pt trilayer and (001)-oriented MgO/CoFeB/Ta trilayer.

## FIG. 6 HERE (Note white space above and below.)

## IV. SUMMARY

Electric field modulations of PMA and the Gilbert damping constant in MgO/Co/Pt trilayers have been investigated by means of TRMOKE and AHE. It was confirmed in both measurements that PMA decreased with increasing electric field where the variation of PMA was -47 fJ/(V·m) for TRMOKE and -32 fJ/(V·m) for AHE. The inverse of the relaxation time of magnetization precession,  $1/\tau$  was also confirmed to be modified by the electric field. Effective damping constant  $\alpha_{eff}$  was estimated from the  $H_{ext}$  dependence of  $1/\tau$  under the application of the electric field from -0.3V/nm to 0.3 V/nm. By considering four contributions to  $1/\tau$ ; effective damping  $\alpha_{\rm eff}$ , anisotropy axis distribution  $\Delta \theta_{\rm H}$ , anisotropy distribution  $\Delta H_k^{\text{eff}}$ , and TMS, the contributions of  $\alpha_{\rm eff}$  and  $\Delta \theta_{\rm H}$  were found to be dominant. Both  $\alpha_{\rm eff}$  and  $\Delta \theta_{\rm H}$ terms decreased with increasing electric field, which is responsible for the electric field modulation of  $1/\tau$ . In the MgO/Co/Pt trilayer,  $\alpha_{eff}$  was found to decrease with increasing electric field with a ratio of -0.0046 nm/V, corresponding to -13 % per 1 V/nm. The modulation of  $\alpha_{\rm eff}$  is comparable to that in previously reported MgO/CoFeB/Ta trilayers. The MgO/CoFeB/Ta trilayers are expected to have MgO(001)/bcc-CoFe interface, which is different structure from our MgO/Co/Pt with MgO(111)/fcc-Co interface. It was found a similar PMA and  $\alpha$  modulations between (111)-orientated MgO/Co/Pt trilayers and (001)-orientated MgO/CoFeB/Ta trilayers.

#### APPENDIX

The external field dependence of the inverse of the relaxation time  $1/\tau$  of the MgO/Co/Pt trilayer under various electric fields was fitted with (3). There were assumed to be four contributions to  $1/\tau$ ;  $\alpha_{\text{eff}}$ ,  $\Delta\theta_{\text{H}}$ ,  $\Delta H_{\text{k}}^{\text{eff}}$  and TMS terms in (3). The TMS term is proportional to the scattering intensity  $N_0$  which given by

$$N_{0} = \gamma^{4} \left( 4H_{X0}^{2} \cos^{4}\theta + 4H_{Y0}^{2} \cos^{2}2\theta - 8H_{X0}H_{Y0} \cos^{2}\theta \cos 2\theta \right) (\Delta H_{k}^{\text{eff}})^{2},$$
(4)

Spin wave dispersion  $\omega_k$  and inverse life time of the spin wave  $\delta \omega_k$  are expressed as

$$\omega_{\mathbf{k}} = \gamma \sqrt{H_X(\mathbf{k}) H_Y(\mathbf{k})},\tag{5}$$

$$\delta\omega_{\mathbf{k}} = \alpha_{\text{eff}}\gamma \left(H_X(\mathbf{k}) + H_Y(\mathbf{k})\right),\tag{6}$$

and  $H_X(\mathbf{k})$  and  $H_Y(\mathbf{k})$  are expressed as

$$H_X(\mathbf{k}) = H_{X0} + \frac{M_{\rm s}}{\mu_0} (1 - N_k) \cos^2 \phi_{\mathbf{k}} + \frac{2A_{\rm ex}}{M_{\rm s}} k^2,$$
  

$$H_Y(\mathbf{k}) = H_{Y0} + \frac{M_{\rm s}}{\mu_0} (1 - N_k)$$
  

$$\times \left\{ -\sin^2 \theta + \cos^2 \theta \sin^2 \phi_{\mathbf{k}} \right\} + \frac{2A_{\rm ex}}{M_{\rm s}} k^2,$$
(7)

where,  $\mu_0$  is the permeability of vacuum,  $\varphi_k$  is the azimuth angle of the spin wave,  $A_{ex}$  is the exchange stiffness and  $N_k$  is wave number dependent demagnetization factor, which is expressed as

$$N_k = \frac{1 - e^{kt_{\rm Co}}}{kt_{\rm Co}},\tag{8}$$

where  $t_{Co}$  is the thickness of Co layer. Correlation function

C(k) is given by

$$C(k) = \frac{2\pi\xi^2}{\left(1 + (k\xi)^2\right)^{3/2}},\tag{9}$$

where  $\xi$  is the correlation length. In this study, we fixed the parameters to simplify the fitting as  $\Delta\theta_{\rm H} = 3.6$  °,  $M_{\rm s} = 1400$  emu/cc,  $\xi = 5$  nm,  $A_{\rm ex} = 26$  pJ/m, since these parameters are considered to be independent of applied electric field.  $\Delta H_{\rm k}^{\rm eff}$  was assumed to be varied with PMA, and set to be 0.8% of perpendicular anisotropy field as,

$$\Delta H_{\rm k}^{\rm eff} = 0.008 \left( H_{\rm k}^{\rm eff} + \frac{M_s}{\mu_0} \right).$$
 (10)

 $H_k^{eff}$  under the electric field was taken from Fig. 3 (b).

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Fig. 1. Top view of a microfabricated MgO/Co/Pt device taken by optical microscope.  $I_{H\pm}$  and  $V_{H\pm}$  indicate the current and voltage terminals for the measurement of anomalous Hall effect, respectively, and  $V_G$  is the terminal for applying the gate voltage.



Fig. 2. XRD profile of ITO(20)/MgO(10)/Co(1.6)/Pt(1.6)/Ta(10)/substrate.The peaks indicated by \* are from the Si substrate.



Fig. 3. (a) Normalized AHE of MgO(10)/Co(1.2)/Pt(1.6) trilayer taken applying in-plane magnetic field  $H_{inp}$  under the application of electric field E = 0.15 V/nm and -0.15 V/nm. (b) Normalized in-plane component of magnetization  $M_{inp}/M_s$ .



Fig. 4. (a) Magnetization dynamics of MgO(10)/Co(1.2)/Pt(1.6) trilayer obtained from TRMOKE measurements under the application of electric field E = 0.3 V/nm and -0.3 V/nm. (b) Electric field dependence of the effective anisotropy field  $H_k^{\text{eff}}$  of the MgO/Co/Pt trilayer obtained by TRMOKE and AHE.



Fig. 5. (a) Electric field dependence of inverse of relaxation time  $1/\tau$  of MgO(10)/Co(1.2)/Pt(1.6) trilayer at an external magnetic field  $H_{\text{ext}} = 14$  kOe. (b)  $H_{\text{ext}}$  dependence of the inverse of relaxation time  $1/\tau$  of the MgO/Co/Pt trilayer under the electric field of E = 0.3 V/nm. The closed circles show the experimental results, which were fit by assuming that  $1/\tau$  consists of four contributions; effective Gilbert damping  $\alpha_{\text{eff}}$ , anisotropy axis distribution  $\Delta \theta_{\text{H}}$ , anisotropy distribution  $\Delta H_{\text{k}}^{\text{eff}}$  and two-magnon scattering (TMS). Each contributions and the sum of these contributions are also shown as solid lines in the figure.



Fig. 6. (a) Electric field dependence of inverse of relaxation time at external magnetic field  $H_{\text{ext}} = 14$  kOe. Experimental values of the inverse of relaxation time  $1/\tau$  of MgO(10)/Co(1.2)/Pt(1.6) trilayer are shown as red circles. Four contributions to  $1/\tau$ : effective Gilbert damping  $\alpha_{\text{eff}}$ , anisotropy axis distribution  $\Delta \theta_{\text{H}}$ , anisotropy distribution  $\Delta H_k$  and two-magnon scattering, are extracted by the fitting of (3). (b) Electric field dependence of the effective damping  $\alpha_{\text{eff}}$  of the MgO/Co/Pt trilayer.