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# Mg-implanted vertical GaN junction barrier Schottky rectifiers with low on resistance, low turn-on voltage, and nearly ideal nondestructive breakdown voltage

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## ABSTRACT

Vertical GaN junction barrier Schottky (JBS) diodes with superior electrical characteristics and nondestructive breakdown were realized using selective-area p-type doping via Mg ion implantation and subsequent ultra-high-pressure annealing. Mg-ion implantation was performed into a 10  $\mu\text{m}$  thick Si-doped GaN drift layer grown on a free-standing n-type GaN substrate. We fabricated the JBS diodes with different n-type GaN channel widths  $L_n = 1$  and 1.5  $\mu\text{m}$ . The JBS diodes, depending on  $L_n$ , exhibited on-resistance ( $R_{ON}$ ) between 0.57 and 0.67  $\text{m}\Omega\text{cm}^2$ , which is a record low value for vertical GaN Schottky barrier diodes (SBDs) and high breakdown (BV) between 660 and 675 V (84.4% of the ideal parallel plane BV). The obtained low  $R_{ON}$  of JBS diodes can be well explained in terms of the  $R_{ON}$  model, which includes n-type GaN channel resistance, spreading current effect, and substrate resistance. The reverse leakage current in JBS diodes was relatively low  $10^3$ – $10^4$  times lower than in GaN SBDs. In addition, the JBS diode with lower  $L_n$  exhibited the leakage current significantly smaller (up to reverse bias 300 V) than in the JBS diode with large  $L_n$ , which was explained in terms of the reduced electric field near the Schottky interface. Furthermore, the JBS diodes showed a very high current density of 5.5  $\text{kA}/\text{cm}^2$ , a low turn-on voltage of 0.74 V, and no destruction against the rapid increase in the reverse current approximately by two orders of magnitude. This work demonstrated that GaN JBS diodes can be strong candidates for low loss power switching applications.

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Vertical gallium nitride (GaN) power diodes are strong candidates for low loss power switching applications due to their low on-resistance ( $R_{ON}$ ) and high breakdown voltage (BV).<sup>1–12</sup> In particular, owing to recent progress in fabrication of a freestanding GaN substrate, it was possible to realize high-quality vertical GaN p–n diodes.<sup>13–19</sup> Despite these excellent results, GaN p–n diodes exhibited a relatively large turn-on voltage ( $V_{ON} \sim 3$ ) originating from the bandgap of GaN, which limited their practical applications due to power loss. On the other hand, Schottky barrier diodes (SBDs) can provide low  $V_{ON}$  (typically  $< 1\text{V}$ ) due to the Schottky barrier interface. Nevertheless, SBDs suffer from the high off-state reverse leakage current due to thermionic-field emission (TFE) at the Schottky interface. The leakage currents of SBDs can be greatly reduced by the application of the junction barrier Schottky (JBS) structure. Additionally, the JBS

structure can also provide an excellent avalanche capability due to the p–n junction, which is important for operating in a switching environment.

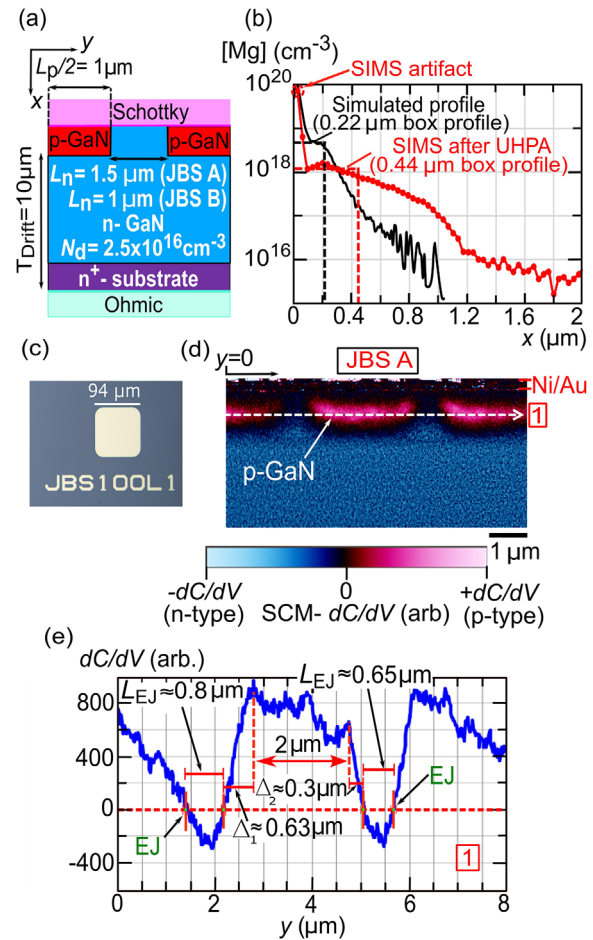
However, realization of the JBS structure for GaN is a challenging issue because of inefficiency of selective-area p-type doping technology. The first vertical Mg-implanted GaN JBS diodes obtained by a multicycle rapid thermal annealing (RTA) method exhibited high BV over 600 V but a very large  $R_{ON}$  ( $> 100\text{ m}\Omega\text{cm}^2$ ).<sup>20</sup> Devices fabricated using this approach were improved by Zhang *et al.*<sup>21</sup> who reported vertical Mg-implanted GaN JBS diodes with the BV of 500–600 V and the  $R_{ON}$  of 1.7  $\text{m}\Omega\text{cm}^2$ . However, the destructive breakdown occurred at the Schottky contact edge, probably due to the low Mg activation ratio. Recently, Fu *et al.*<sup>22</sup> reported the avalanche capability in the Mg-implanted GaN JBS diodes obtained by the multicycle RTA method,

but  $R_{ON}$  was still high ( $\sim 3 \text{ m}\Omega \text{ cm}^2$ ) as for the BV of 600–800 V. Due to the lack of effective Mg implantation technology, the other processes or structures were employed to GaN SBDs such as the vertical regrown JBS (RJBS),<sup>23</sup> trench JBS structure (TJBS),<sup>24,25</sup> or trench MOS barrier Schottky (TMBS) structure.<sup>26,27</sup> They exhibit very interesting properties, like keeping low leakage currents at high temperatures (TMBS structures), but they need the specific design and/or complex technological processes. On the other hand, Mg-ion implantation technology can offer flexible/simplified device design and reduced device fabrication complexity.

Compared to silicon carbide (SiC) JBS rectifiers, the GaN JBS diodes represent an early stage of development. Thus, the ability to realize high performance vertical GaN JBS diodes can bring GaN power electronics to the next level. Recently, our group fabricated various Mg-implanted GaN based edge termination structures<sup>28,29</sup> using an ultra high-pressure annealing (UHPA) method,<sup>30–33</sup> which ensured a high Mg activation ratio. In this Letter, we fabricated the Mg-implanted GaN vertical JBS diodes using the UHPA process and demonstrated their superior electrical characteristics and nondestructive breakdown.

Figure 1(a) shows the schematic cross-sectional image of the Mg-implanted vertical GaN JBS diodes. These devices were fabricated using a  $10 \mu\text{m}$  thick silicon (Si)-doped GaN drift layer grown on free-standing GaN (0001) substrates prepared via hydride vapor phase epitaxy. The threading dislocation density was relatively low of the order of  $10^6 \text{ cm}^{-2}$ . The effective donor concentration ( $N_d$ ) of the drift layer was estimated to be  $\approx 2.5 \times 10^{16} \text{ cm}^{-3}$  from the capacitance–voltage measurements. The JBS diodes with different n-GaN channel widths ( $L_n$ ) were designed [see Fig. 1(a)]:  $L_n = 1.5 \mu\text{m}$  (JBS A) and  $L_n = 1.0 \mu\text{m}$  (JBS B). The width of the implanted p-type region ( $L_p$ ) for both JBS diodes was designed to be  $2 \mu\text{m}$ . The thickness of the drift layer was chosen as  $10 \mu\text{m}$  in order to avoid the punch-through phenomenon (according to Ref. 34, for  $N_d = 2.5 \times 10^{16} \text{ cm}^{-3}$ , the non-punch-through drift layer should have the thickness larger than  $5 \mu\text{m}$ ).

The first step of the fabrication process was etching of an  $1 \mu\text{m}$  thick  $\text{SiO}_2$  mask in order to prepare the chosen regions for the implantation. Subsequently, the Mg ions were implanted with the energy of 20 and 180 keV at the respective dosages of  $5 \times 10^{14}$  and  $8 \times 10^{13} \text{ cm}^{-2}$  at a tilt angle of  $7^\circ$ . Such a combination of the Mg energy and dosages leads to creation of a high  $p^+$  region near the surface with the Mg concentration ( $[\text{Mg}]$ ) of  $10^{20} \text{ cm}^{-3}$  and  $0.22 \mu\text{m}$  thick box profile<sup>35</sup> with the average  $[\text{Mg}]$  of  $4.5 \times 10^{18} \text{ cm}^{-3}$  according to the Monte Carlo (MC) simulations<sup>36</sup> [see Fig. 1(b)]. After implantation and mask removal, the UHPA process was conducted under an  $N_2$  pressure of 500 MPa at a temperature of  $1300^\circ\text{C}$  for 30 min at Japan Ultra-High Temperature Materials Research Institute in Ube City, Yamaguchi Prefecture, Japan. The Mg depth profile estimated by SIMS after the UHPA process has been shown in Fig. 1(b). The experimental Mg-depth profile extended more deeply in GaN than those predicted by MC simulations due to Mg diffusion,<sup>37,38</sup> which occurred during the UHPA process. The estimated width of the experimental Mg box profile was  $0.44 \mu\text{m}$  with the average  $[\text{Mg}]$  of  $1.18 \times 10^{18} \text{ cm}^{-3}$  [Fig. 1(b)]. After the UHPA process, the Ti/Al/Ni/Au Ohmic contacts were deposited on the substrate at  $475^\circ\text{C}$ , and in the last step, the Ni/Au Schottky contacts were formed on the top. The JBS diodes were formed in a square shape with rounded corners as can be seen in the optical image shown in Fig. 1(c). The device active area contained



**FIG. 1.** (a) Schematic cross section of the JBS diodes, (b) SIMS depth profile of  $[\text{Mg}]$  together with the simulated  $[\text{Mg}]$  depth-profile, (c) optical image of the fabricated JBS diode, (d) SCM image of JBS A, and (e) linear profile of  $dC/dV$  obtained along line 1 from the SCM image of JBS A [Fig. 1(d)]. The high  $[\text{Mg}]$  peak on the surface [Fig. 1(b)] is probably due to a SIMS artifact.

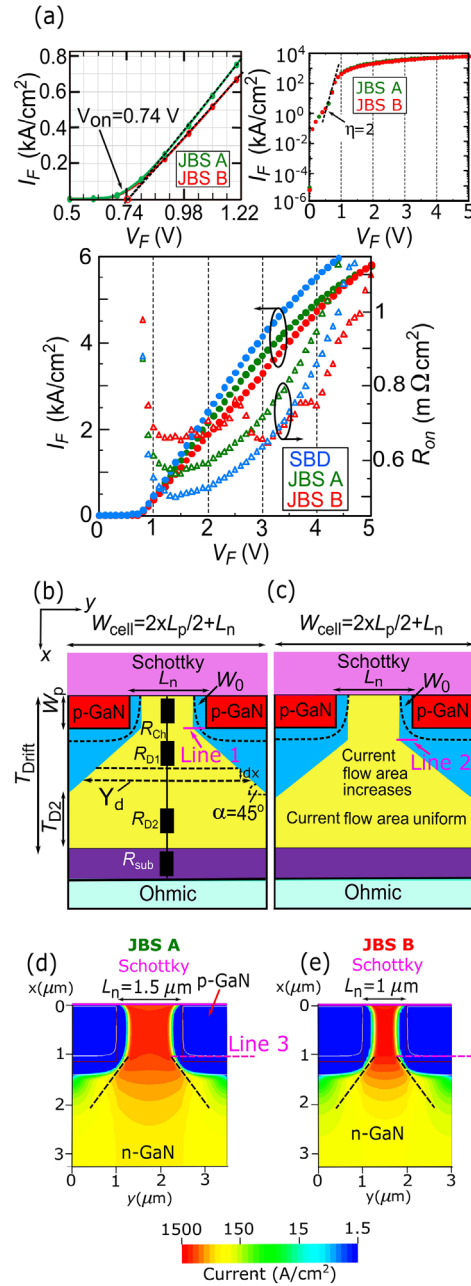
periodically placed n-type and p-type regions, whose numbers of repetitions were 17 and 20 for JBS A and B, respectively.

The strong contrast between p-type and n-type regions was observed at the cross-sectional scanning capacitance microscopy (SCM) image [see Fig. 1(d)], which indicates a conversion of the n-type GaN region into p-type one. Figure 1(e) shows the linear profile of  $dC/dV$  along line 1 from the SCM image of JBS A [Fig. 1(d)]. The electrical junction (EJ) locations,<sup>39</sup> i.e., points where the free electron and hole concentrations are equal (or points where the Fermi level is located at the midgap) can be clearly recognized. For JBS A, according to the TCAD simulations (see the supplementary material), the distance between EJ points ( $L_{EJ}$ ) should be  $1.2 \mu\text{m}$ , which is larger than experimental  $L_{EJ} = 0.65\text{--}0.8 \mu\text{m}$  [Fig. 1(e)]. This discrepancy can be due to the UHPA Mg lateral diffusion, which leads to shrinking of  $L_n$  (and expanding  $L_p$ ) and/or not accurate EJ location estimation in the SCM measurements. In particular, it was shown that the EJ location in SCM measurements can be moved due to the AC tip bias.<sup>39–41</sup>

To estimate the EJ error location, we compared two regions  $\Delta_1$  and  $\Delta_2$  [see Fig. 1(e)], which roughly reflect expanding of  $L_p$  from the original size  $2\ \mu\text{m}$ . If the Mg lateral diffusion would be only responsible for reducing  $L_{EJ}$ ,  $\Delta_1 = \Delta_2$ . (Mg diffusion is expected to be similar in both directions.) In our case,  $\Delta_1 \gg \Delta_2$ , which is likely due to the influence of the AC tip bias on the EJ location. Thus, the error of the EJ location can be estimated as:  $\frac{\Delta_1 - \Delta_2}{2} \approx 0.165\ \mu\text{m}$ . This means that the true  $L_{EJ}$  is  $L_{EJ} = 0.98\ \mu\text{m}$  ( $0.65 + 2 \times 0.165$ ) and  $L_{EJ} = 1.13\ \mu\text{m}$  ( $0.8 + 2 \times 0.165$ ). (We assumed that the true  $L_{EJ}$  should be larger than that obtained from the measurement because the AC tip bias enhances the depletion region.<sup>40</sup>) However, the determined  $L_{EJ}$  is still smaller than theoretical  $L_{EJ} = 1.2\ \mu\text{m}$ , which indicates the presence of Mg lateral diffusion with a range  $L_R$  between 0.11 and  $0.035\ \mu\text{m}$ . Due to the Mg lateral diffusion,  $L_n$  is reduced by  $2 \times L_R$  ( $L_p$  expands by  $2 \times L_R$ ), and it is equal to  $1.28\text{--}1.43$  and  $0.78\text{--}0.93\ \mu\text{m}$  for JBS A and B, respectively.

The room temperature forward ( $I_F\text{--}V_F$ ) and reverse ( $I_R\text{--}V_R$ ) current-voltage characteristics were carried out using an Agilent B1505A semiconductor analyzer. The representative linear-scale  $I_F\text{--}V_F$  and the differential  $R_{ON}\text{--}V_F$  characteristics of JBS A and B as well as the Schottky barrier diode (SBD) are shown in Fig. 2(a). Note that  $I_F$  in Fig. 2(a) is normalized with respect to the total device active area (Schottky electrode area) equal to  $8.6 \times 10^{-5}\ \text{cm}^2$ . Unfortunately, the implanted pn diodes exhibited very low  $I_F$  due to the poor Ohmic contacts to implanted p-GaN, and thus, they were not shown in Fig. 2(a). The reason for poor Ohmic contacts to implanted p-GaN is likely low Mg concentration near the surface after the UHPA process [see Fig. 1(b)]. At  $V_F = 1.5\text{--}1.6\ \text{V}$ , the JBS A and B exhibited the minimum  $R_{ON}$  of 0.57 and  $0.67\ \text{m}\Omega\ \text{cm}^2$ , respectively, while for SBD,  $R_{ON}$  was  $0.51\ \text{m}\Omega\ \text{cm}^2$ . In addition, from Fig. 2(a) and its inset, one can note that the JBS diodes exhibit a very high  $I_F > 5.5\ \text{kA/cm}^2$  at 5 V (the total current at 5 V was 0.5 A) and low  $V_{ON}$  of 0.74 V. However, the ideality factor ( $\eta$ ) of our diodes was relatively high, around 2 [see the inset of Fig. 2(a)] probably due to the defects induced by the UHPA process.

The  $R_{ON}$  value of JBS B was higher than that of JBS A in the range  $V_F$  up to 2.5 V [see Fig. 2(a)], probably due to the shorter  $L_n$  and, thus, the higher vertical channel resistance ( $R_{Ch}$ ). In the range  $V_F \approx 2.5\text{--}3\ \text{V}$ ,  $R_{ON}$  of the JBS B decreases to the level of the SBD diode [Fig. 2(a)]. This suggests that some of the metal p-GaN contacts in the JBS B could be rather Ohmic-like but not Schottky ones because the range of  $V_F$  from 2.5 to 3 V corresponds to the turn-on voltage of GaN pn diodes. However, as we mentioned previously, the Ohmic contacts were poor. Thus, the reason for this issue is rather not clear at this moment. Another interesting problem is decreasing of the slope of  $I_F\text{--}V_F$  curves above  $V_F > 3\ \text{V}$  for all diodes (an increase in  $R_{ON}$ ), as can be seen in Fig. 2(a). Previously, such a phenomenon was observed for SiC and  $\text{Ga}_2\text{O}_3$  SBD/JBS diodes<sup>42,43</sup> and attributed to the self-heating effect (Joule heat generation). More precisely, when high  $I_F$  is reached the SBD diode (in the case of the SiC SBD, it is of the order of magnitude of  $\text{kA/cm}^2$ ), the Joule heat is generated, which leads to a significant increase in the lattice temperature and large degradation in the carriers mobility. As a consequence,  $R_{ON}$  increases and the slope of the  $I_F\text{--}V_F$  curve decreases, as shown in Fig. 2(a). However, in the case of GaN based SBD diodes, the decrease in slope of the  $I_F\text{--}V_F$  curve is not often observed for  $V_F$  up to 5 V, and thus, the question arises why this phenomenon occurred in the fabricated diodes. The simple answer is that this is a direct consequence of low  $R_{ON}$  of our diodes. Namely, due to low  $R_{ON}$ , the amount of Joule heat ( $Q = V_F^2 t / R_{ON}$ ,



**FIG. 2.** (a) Linear-scale  $I_F\text{--}V_F$  curves of SBD and JBS diodes and corresponding  $R_{ON}\text{--}V_F$  characteristics. Current distributions in the JBS diode: (b) the start of current spreading from the end of the pn junction (line 1) and (c) the start of current spreading from the end of the depletion region (line 2). Two-dimensional (2D) current density distribution in (d) JBS A and (e) JBS B at  $V_F = 1\ \text{V}$ . Inset of Fig. 2a shows the linear-scale  $I_F\text{--}V_F$  characteristics at  $V_F$  range from 0.5 to 1.22 V and semilog  $I_F\text{--}V_F$  curves of JBS diodes.

where  $t$  is the time) generated in our diodes starts to be significant already at relatively low  $V_F$ , which causes degradation in the carrier mobility the carriers mobility and the decrease in the slope of the  $I_F\text{--}V_F$  curve.

The difference between  $R_{ON}$  of JBS diodes and SBD ones can be explained using the following  $R_{ON}$  model. The  $R_{ON}$  of the JBS diode contains the series of various resistances, as shown in Fig. 2(b) and can be given by

$$R_{ON} = R_{Ch} + R_{Drift} + R_{Sub} + R_{Con}, \quad (1)$$

where  $R_{Drift}$ ,  $R_{Sub}$ , and  $R_{Con}$  are the drift layer resistance, substrate resistance, and contact resistance, respectively. The  $R_{Con}$  is typically much lower than  $R_{Sub}$  and, thus, can be neglected. For the calculations of JBS  $R_{ON}$ , two scenarios of the current distribution can be adopted. In the first one, the current spreading starts from the end of the pn junction [line 1 in Fig. 2(b)] (similar like in VD-MOSFET, see Ref. 44) while in the second scenario, the current spreading starts from the end of the depletion region [see line 2 in Fig. 2(c)] (similar like in SiC JBS diodes, see Refs. 45, 46). According to the TCAD simulation, in the case of the fabricated JBS diodes, the current spreading starts from line 3 [see Figs. 2(d) and 2(e)], which is close to line 1 [see Fig. 2(b)]. This means that the current distribution from Fig. 2(b) is more adequate to our JBS diodes, and thus,  $R_{Ch}$  can be calculated similarly like the JFET resistance in VD-MOSFET [see Eq. (6.72) in Ref. 44]

$$R_{Ch} = \frac{\rho W_{Cell} W_p}{L_n - 2 \times W_0}, \quad (2)$$

where  $\rho$  is the resistivity of the n-GaN region,  $W_{Cell} = 2 \times L_p/2 + L_n$  is the cell width,  $W_p$  is the p-n junction depth, and  $W_0$  is the depletion region width at 0 V bias [Fig. 2(b)]. The  $R_{Drift}$  in the case of the JBS structure is expected to be different than in the case of SBD due to the current spreading from the n-GaN channel, as shown in Fig. 2(c). Assuming the  $45^\circ$  spreading angle,  $R_{Drift}$  of the JBS diode is determined by the region where the current flow area increases [Fig. 2(c)] and the region where the current flow area is uniform [Fig. 2(c)]. The relationship for the resistance of a layer, in which the current flow is non-uniform,  $R_{D1}$  is obtained by the integration of the resistance of an elemental segment with thickness  $dx$  and width  $Yd = a + 2x$  [see Fig. 2(b)] over the limits  $x = 0$  and  $x = (W_{Cell} - a)/2$ , where  $a = L_n - 2 \times W_0$

$$R_{D1} = \int dR_{D1} = \int W_{Cell} \rho \frac{dy}{a + 2x} = \frac{\rho W_{Cell}}{2} \ln \left[ \frac{W_{Cell}}{L_n - 2 \times W_0} \right]. \quad (3)$$

Note that the above equation was multiplied by the cell area.

On the other hand, the resistance of the layer in which the current flow is uniform,  $R_{D2}$  is determined by the calculation of the thickness of this layer [ $T_{D2}$ , see Fig. 2(b)]. By the simple consideration of a  $45^\circ$  triangle from Fig. 2(b), we obtain that  $T_{D2}$  is equal

$$T_{D2} = T_{Drift} + \frac{L_n}{2} - W_0 - \frac{W_{Cell}}{2}, \quad (4)$$

where  $T_{Drift}$  is the drift layer thickness. Thus,  $R_{D2}$  is given by

$$R_{D2} = \rho L_{D2} = \rho \left[ T_{Drift} + \frac{L_n}{2} - W_0 - \frac{W_{Cell}}{2} \right]. \quad (5)$$

Finally,  $R_{Drift}$  is the sum of  $R_{D1}$  and  $R_{D2}$ ,

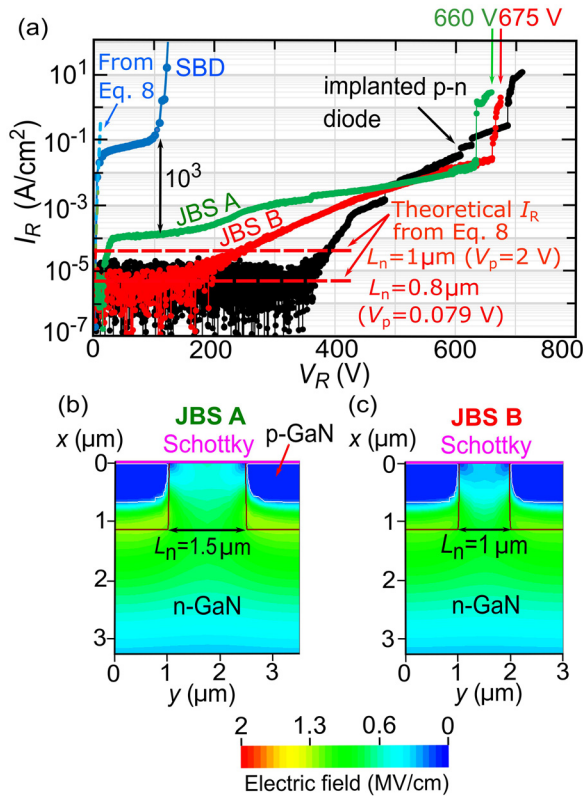
$$R_{Drift} = R_{D1} + R_{D2} = \frac{\rho W_{Cell}}{2} \ln \left[ \frac{W_{Cell}}{L_n - 2 \times W_0} \right] + \rho \left[ T_{Drift} + \frac{L_n}{2} - W_0 - \frac{W_{Cell}}{2} \right]. \quad (6)$$

All the parameters in Eqs. (1)–(6) are known except from  $\rho$  which can be determined from  $R_{ON}$  of SBD. Namely,  $R_{ON}$  of SBD is given by the relationship (after neglecting  $R_{Con}$ )

$$R_{ON} = R_{DS} + R_{Sub}, \quad (7)$$

where  $R_{DS} = \rho T_{Drift}$  is the drift layer resistance of SBD. The  $R_S$  value for the used GaN substrate was estimated as  $0.294 \pm 0.023 \text{ m}\Omega \text{ cm}^2$ , and thus from Eq. (7), we obtained  $\rho = 217 \pm 23 \text{ m}\Omega \text{ cm}$  (assuming  $R_{ON} = 0.51 \text{ m}\Omega \text{ cm}^2$  of SBD). The  $\rho$  range of 194–240  $\text{m}\Omega \text{ cm}$  corresponds to the average mobility  $\mu = 1/(qN_D\rho) = 1166 \pm 124 \text{ cm}^2/\text{Vs}$ , which is reasonable for the GaN layer with  $N_D = 2.5 \times 10^{16} \text{ cm}^{-3}$  grown on a GaN substrate.<sup>47</sup> In order to improve the accuracy of the  $R_{Ch}$  estimations, we used the  $L_n$  determined based on SCM data [see discussion on Figs. 1(d) and 1(e)]. For the JBS A, average  $L_n \approx 1.36$  and  $W_{Cell} = 3.5 \mu\text{m}$ , while for JBS B, average  $L_n \approx 0.86$  and  $W_{Cell} = 3 \mu\text{m}$ . Thus, after introducing these parameters to Eqs. (1)–(6) together with  $\rho = 220 \text{ m}\Omega \text{ cm}$ ,  $T_{Drift} = 10 \mu\text{m}$ , and  $W_0 = 0.31 \mu\text{m}$  (calculated assuming the graded p-n junction, see the supplementary material), and  $W_p = 1.1 \mu\text{m}$  [see the SIMS profile, Fig. 1(b), point where  $[\text{Mg}] = N_D$ ], we obtained that  $R_{ON}$  of the JBS A and B is  $\approx 0.64$  and  $0.86 \text{ m}\Omega \text{ cm}^2$ , respectively. These values are in good agreement with the experimental ones ( $0.57$  and  $0.67 \text{ m}\Omega \text{ cm}^2$ ). Based on these results, we can conclude that the increases in  $R_{ON}$  of both JBS diodes with respect to SBD are mainly due to  $R_{Ch}$ . The difference between the experimental and theoretical  $R_{ON}$  values is probably due to overestimation of the Mg lateral diffusion from the SCM analysis [Fig. 1(e)] and not accurate estimation of  $\rho$  (194–240  $\text{m}\Omega \text{ cm}$ ). If we assume the lack of Mg lateral diffusion, i.e.,  $L_n$  values as designed [see Fig. 1(a)] and  $\rho = 194 \text{ m}\Omega \text{ cm}$ , we obtain the  $R_{ON}$  values almost the same as experimental ones, i.e.,  $0.59 \text{ m}\Omega \text{ cm}^2$  (JBS A) and  $0.69 \text{ m}\Omega \text{ cm}^2$  (JBS B). These results suggest that the EJ locations [see Fig. 1(e)] are mainly affected by the AC tip bias.

Figure 3(a) shows the representative  $I_R$ - $V_R$  characteristics of the JBS diodes, SBDs, and implanted p-n diodes. The  $I_R$ - $V_R$  were obtained using the fluorinert. One can note the large improvement of the reverse characteristics of the JBS diodes with respect to SBD. In particular, the SBD exhibited the BV of 120 V while the JBS diodes BV of 660–675 V (84.4% of ideal BV = 800 V for  $N_D = 2.5 \times 10^{16} \text{ cm}^{-3}$ ; more details on the ideal BV, one can find in Ref. 34), which is close to the implanted p-n diode. The maximal electric field ( $E_{max}$ ) at the breakdown (estimated according to Ref. 13) was  $E_{max} > 2.47 \text{ MV/cm}$  (for JBS A), which is consistent with  $E_{max}$  reported by Maeda *et al.*<sup>34,48,49</sup> for  $N_D = 2.5 \times 10^{16} \text{ cm}^{-3}$ . Furthermore, compared to SBDs, the JBS devices had much lower leakage currents (by the factor of  $10^3$ – $10^4$  orders of magnitude) at  $V_R = 120 \text{ V}$ . It is interesting also to note that the leakage currents in the JBS B are one order of magnitude lower than in JBS A. Moreover, the behavior of the leakage currents as a function of  $V_R$  is slightly different between the JBS diodes A and B. The leakage currents in the JBS A initially increase like in SBD while in the JBS B such an initial increase is not observed [Fig. 3(a)]. To understand this discrepancy, we simulated the two-dimensional (2D) electric field distribution in the JBS A and B, as shown in Figs. 3(b) and 3(c) under  $V_R = 200 \text{ V}$ . It is clear that in the case of the JBS B, the electric field in the n-GaN channel near the Schottky interface is significantly lower than in the JBS A. This is because in the JBS B, the p-type regions are much closer than in the JBS A with more effectively depleted n-GaN channel and, thus, reduced electric field near the



**FIG. 3.** (a)  $I_R$ - $V_R$  characteristics of SBD, JBS, and p-n diodes and calculated 2D electric field distributions in JBS A (b) and (c) B at  $V_R = 200 \text{ V}$ . Dashed lines were calculated using Eq. (8).

Schottky interface. Due to the lower electric field near the Schottky interface, TFE in the JBS B is more suppressed compared to the JBS A, which results in the lower leakage currents and weaker  $V_R$  dependencies. An additional finding in Fig. 3(a) is that the pn diode exhibits the higher leakage current than JBS diodes at bias larger than 500 V. We attributed this phenomenon to the Mg condensation at threading dislocations<sup>47</sup> (TSDs), which is more probable in the pn diodes than in JBS ones because of much larger implanted Mg areas. Finally, it is important to note that the  $I_R$ - $V_R$  characteristics of our JBS diodes can be excellently explained using a well verified reverse leakage model of the JBS diode.<sup>45,46</sup> According to this model,  $I_R$  of the JBS diode is given by

$$I_R = \frac{L_n}{W_{Cell}} A_R T^2 \exp\left(-\frac{q\Phi}{kT}\right) \exp\left(\frac{q^3 E_{JBS}^{\frac{1}{2}}}{(4\pi\epsilon_s)^{\frac{1}{2}} kT}\right) \exp(CE_{JBS}^2), \quad (8)$$

where  $A_R$  is the Richardson constant equal to  $24 \text{ A}/(\text{K}^2 \text{ cm}^2)$ ,  $\epsilon_s$  is the dielectric constant of GaN with the relative permittivity of 10.4,  $C$  is the tunneling constant,  $\Phi$  is the SBD barrier height, and  $E_{JBS}$  is the electric field in the Schottky region given by

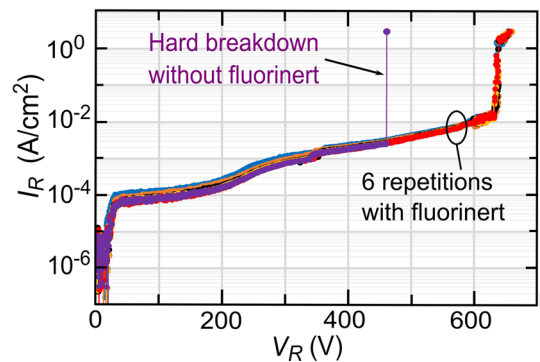
$$E_{JBS} = \sqrt{\frac{2qN_d}{\epsilon_s} (V_R + V_b)}, \quad (9)$$

where  $V_b$  is the Schottky contact potential.

When  $V_R$  exceeds the pinch-off voltage  $V_P = \frac{L_n^2 q N_d}{8\epsilon_s} - V_B$  [where  $V_B$  is the built-in voltage of the p-n junction (3.4 V)],  $E_{JBS}$  and, thus,  $I_R$  of JBS become approximately constant due to the channel pinch-off effect. From fitting of the experimental  $I_R$  for a simple SBD in Eq. (8) [see Fig. 3(a)] (without the  $L_n/W_{Cell}$  factor), we determined  $\Phi = 0.7 \text{ V}$  and  $C = 10^{-10} \text{ cm}^2/\text{V}^2$ . We used these parameters to calculate  $I_R$  of JBS diodes using Eq. (8). As can be seen from Fig. 3(a), when we assume  $L_n$  values very close to the designed ones, i.e.,  $0.8 \mu\text{m}$  ( $V_P = 0.079 \text{ V}$ ) for JBS diode B, the calculated  $I_R$  is in excellent agreement with experimental data. Furthermore, for JBS B, if we assume the exact designed  $L_n$  value of  $1 \mu\text{m}$  ( $V_P = 2 \text{ V}$ ),  $I_R$  is also consistent with experimental one [see Fig. 3(a)]. This means that  $I_R$  of the JBS diodes can be well explained only by variations of the  $L_n$  parameter.

Figure 4 shows the  $I_R$ - $V_R$  characteristics of JBS A after several repetitive measurements. The diode showed no destruction against the rapid increase in  $I_R$  above two orders of magnitude up to  $3 \text{ A}/\text{cm}^2$ . Unfortunately, the temperature dependencies of the  $I_R$ - $V_R$  characteristics were difficult to obtain since BV of the JBS diodes was very sensitive to the presence of the fluorinert medium, which evaporates at elevated temperatures (without the fluorinert medium, the BV of JBS diodes was below 500 V, see Fig. 4). Figure 5 shows the dependencies of  $R_{ON}$  vs BV of GaN-based quasi and vertical SBDs and GaN p-n diodes with BV of 700 V. The figure of merit of the fabricated JBS diodes was from 0.68 to  $0.76 \text{ GW}/\text{cm}^2$ , which is one of the highest values reported so far for GaN SBDs. Furthermore,  $R_{ON}$  of our JBS diodes is the lowest among all reported so far for the vertical GaN SBDs and is comparable with  $R_{ON}$  of vertical GaN p-n diodes with the BV of 630–720 V. It should also be highlighted that  $R_{ON}$  of JBS diodes will still be a record low even if  $I_F$  of JBS diodes is normalized to the total JBS device area of  $9.7 \times 10^{-5} \text{ cm}^2$  (i.e., the Schottky metal area plus edge termination area). In this case,  $R_{ON}$  will be  $0.64 \text{ m}\Omega \text{ cm}^2$  for JBS A and  $0.75 \text{ m}\Omega \text{ cm}^2$  for JBS B.

In conclusion, we demonstrated the Mg-implanted GaN vertical JBS diodes fabricated using the UHPA method, which exhibit superior electrical characteristics and nondestructive breakdown. The JBS diodes have the record low  $R_{ON}$  in the range from 0.57 to  $0.67 \text{ m}\Omega \text{ cm}^2$ , high BV in the range from 660 to 675 V and low  $V_{ON}$  of 0.74 V. In addition, the JBS diodes showed a very high  $I_F > 5 \text{ kA}/\text{cm}^2$  and relatively low leakage currents. This work showed that GaN JBS diodes can be strong candidates for low loss power switching applications.



**FIG. 4.**  $I_R$ - $V_R$  characteristics of the JBS diode after several-time repeated treatment using the fluorinert and  $I_R$ - $V_R$  characteristics of the same diode obtained without fluorinert.

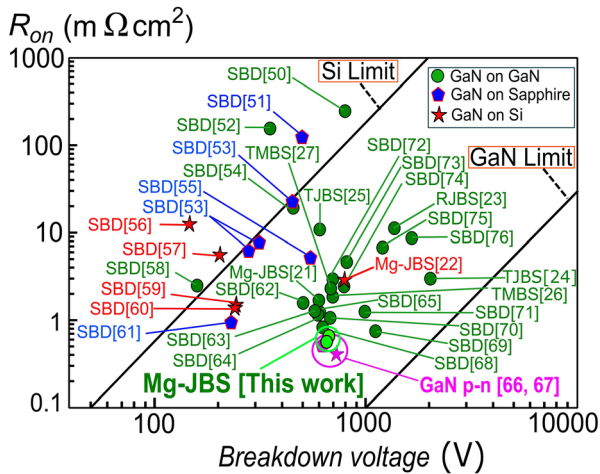


FIG. 5.  $R_{ON}$ -BV benchmark comparison among most GaN SBDs,<sup>50–65,68–76</sup> Mg-implanted JBS,<sup>21,22</sup> regrowth JBS (RJBS),<sup>23</sup> trench JBS (TJBS),<sup>24,25</sup> TMBS,<sup>26,27</sup> and p-n diodes (PNDs)<sup>66,67</sup> with the BV of 630–720 V. Our JBS diodes demonstrate the lowest  $R_{ON}$  for vertical SBDs, which are comparable with  $R_{ON}$  of GaN p-n diodes.

See the [supplementary material](#) for estimations of the electrical junction location (Fig. S1) and depletion layer width at the zero bias (Fig. S2) from TCAD simulations.

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## AUTHOR DECLARATIONS

### Conflict of Interest

The authors have no conflicts to disclose.

### Author Contributions

**Maciej Franciszek Matys:** Conceptualization (lead); Data curation (equal); Formal analysis (lead); Investigation (equal); Methodology (equal); Software (lead); Validation (equal); Visualization (equal); Writing – original draft (lead); Writing – review & editing (lead). **Kazuki Kitagawa:** Conceptualization (supporting); Data curation (supporting); Formal analysis (supporting); Investigation (supporting); Methodology (supporting); Resources (supporting); Software (equal); Validation (supporting); Visualization (supporting); Writing – original draft (supporting); Writing – review & editing (supporting). **Tetsuo**

**Narita:** Conceptualization (equal); Data curation (equal); Formal analysis (equal); Investigation (equal); Methodology (equal); Resources (equal); Validation (equal); Visualization (equal); Writing – original draft (equal); Writing – review & editing (equal). **Tsutomu Uesugi:** Conceptualization (equal); Data curation (equal); Formal analysis (equal); Investigation (equal); Methodology (equal); Resources (equal); Validation (equal); Visualization (equal); Writing – original draft (equal); Writing – review & editing (equal). **Jun Suda:** Conceptualization (equal); Data curation (equal); Formal analysis (equal); Funding acquisition (equal); Investigation (equal); Methodology (equal); Project administration (lead); Resources (equal); Supervision (lead); Validation (equal); Visualization (equal); Writing – original draft (equal); Writing – review & editing (equal). **Tetsuo Kachi:** Conceptualization (lead); Data curation (equal); Formal analysis (equal); Funding acquisition (equal); Investigation (equal); Methodology (equal); Project administration (lead); Resources (equal); Supervision (lead); Validation (equal); Visualization (equal); Writing – original draft (equal); Writing – review & editing (equal).

## DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding author upon reasonable request.

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