

# Effect of Nanorod Alignment on Flux Pinning State in BaHfO<sub>3</sub> Doped SmBa<sub>2</sub>Cu<sub>3</sub>O<sub>y</sub> Films

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**Abstract**—The effect of the microstructure on the flux pinning state in BaHfO<sub>3</sub>(BHO) nanorods doped SmBa<sub>2</sub>Cu<sub>3</sub>O<sub>y</sub>(SmBCO) films was investigated using high field magnets with a rotator stage. A films with aligned nanorods showed the strong flux pinning along *c*-axis, which disappears at high magnetic fields. This behavior corresponds to the existence of the Bose glass state in the film which is also supported by the behaviors of the activation energy and the critical exponent of the liquid-glass transition estimated from the transport properties. Moreover, the field dependences of the flux pinning force are explained by the flux pinning models for the correlated nanorods and the random pinning centers in the films with the aligned and the firework nanorods structures. The strong relationship is confirmed between the structure of the nanorods and the pinning properties in the BHO doped SmBCO films.

**Index Terms**— High temperature superconductors, Superconducting Films, Flux pinning, Critical current density, Nanostructured materials.

## I. INTRODUCTION

FLUX pinning is the key to enhance the current capacity of the high temperature superconducting wires and tapes for the high temperature and high magnetic field applications because the critical current rapidly decreases at high temperatures due to thermally fluctuated vortex motion [1]–[4]. To overcome the problem of the operation at high temperatures, it is necessary to introduce pinning centers intentionally into the wires and tapes, which is known as artificial pinning centers (APCs) [5]–[7]. Among the high temperature superconductors, REBa<sub>2</sub>Cu<sub>3</sub>O<sub>y</sub> (REBCO, RE = rare earth elements and Y) tapes are the strong candidate for the applications at high temperatures and the high magnetic field at present. To enhance the flux pinning at high temperatures in REBCO tapes, self-organized nanorods have been enthusiastically studied over ten years. Especially, BaMO<sub>3</sub> (BMO) materials forms narrow nanorods and extremely enhance flux pinning in the PLD or the CVD processed REBCO tapes [8]–[10]. According to the material researches, M = Sn, Hf, Zr are the elements to form the nanorods. For further improvements, it is necessary to discuss the vortex pinning mechanisms and the microstructure from the physical view point. The shape of the BMO nanorods depends on the growth conditions especially on temperature [11]–[14]. In recent years, our group developed a growth technique to control the shape of

the nanorod, where the REBCO seed layer is deposited at usual temperature before fabrication of the nanorod doped REBCO film at relatively low temperature to extend the growth temperature range without appearance of the impurity such as *a*-axis oriented grains. We call the 2 step fabrication method as the “low temperature growth (LTG)” technique. According to the microstructure analysis, nanorods tend to become narrow, discontinuous, and inclined with the LTG technique. In this work, we investigate the effect of the nanorod alignments on the vortex pinning state in SmBa<sub>2</sub>Cu<sub>3</sub>O<sub>y</sub> (SmBCO) films with the BaHfO<sub>3</sub> (BHO) nanorods by controlling the nanorod alignment using the LTG technique.

## II. EXPERIMENTAL METHOD

SmBCO films with BHO nanorods were fabricated on LaAlO<sub>3</sub> (001) substrates using the alternating-target pulse laser deposition method with KrF excimer laser (248 nm wavelength) [12]–[14]. The targets were SmBa<sub>2</sub>Cu<sub>3</sub>O<sub>y</sub> and BaHfO<sub>3</sub>. The LTG technique was adopted to incline the nanorods. The ~50 nm thick SmBCO seed layer was firstly deposited at 840 °C and then the ~300 nm thick BHO doped SmBCO upper layer was deposited at 750 °C with the laser pulse frequency of 10 Hz and with the laser pulse energies of 52 and 53 mJ. The BHO content is fixed as 1.6vol.%. Figure 1 shows TEM images of two samples with aligned and inclined nanorods with the smaller and the larger energies of the laser pulses, respectively. We put names of the samples as “Aligned” and “Firework” films. According to the resistivity measurement, *T<sub>c</sub>* of the Aligned and the Firework films are 90.3 K and 91.9 K, respectively. In the Aligned film, nanorods distributes uniformly as shown in Fig. 1(a) and the nanorods are aligned as shown in Fig. 1(b). The nanorods alignment is quantitatively analyzed in the following part. On the contrary in the Firework film, the nanorods are located radially as shown in Fig. 1(c) and are inclined against the *c*-axis of the SmBCO matrix as shown in Fig. 1(d). The nanorod density is calculated as 1000 μm<sup>-2</sup> in the Aligned film as 2 T, but not for the Firework film because of the inhomogeneous distribution of the nanorods. Transport properties including the critical current density *J<sub>c</sub>* and the resistivity *ρ* were measured in high magnetic fields at wide range of temperatures using the 20 T superconducting magnets at HFLSM, IMR, Tohoku Univ. The resistivity is measured

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with the fixed current density of  $10 \text{ A/cm}^2$  and  $J_c$  has the fixed electric field criterion of  $1 \text{ } \mu\text{V/cm}$  and the irreversible magnetic field  $B_{\text{irr}}$  is defined with a criterion of the  $J_c$  of  $10 \text{ A/cm}^2$ .  $B_{\text{irr}}$  is estimated with the same criterion of  $J_c$  based on interpolation between the measured points of different field and temperatures.

Fig. 1 Here

### III. RESULTS AND DISCUSSION

First, the TEM images are analyzed with distribution of the inclined angle against  $c$ -axis of the SmBCO matrix. Figure 2 shows the histogram of the nanorods angle from  $c$ -axis of the films. The distribution is apparently wide for the Firework films. According to the planer view of the TEM images as shown in Fig.1(a) and 1(c), the nanorods looks like circle and oval in the Aligned and Firework films, respectively, because of the angle between the nanorods and the in-plane. The angle distribution is fitted with the Gaussian curves and the characteristic angle distributions are calculated. The Aligned and the Firework films have the characteristic angle of  $8.4^\circ$  and  $24.1^\circ$ , respectively, which are relatively large values compared with the ones for the conventional PLD method [12]-[14]. The angle distribution of the nanorods is related with the perpendicular growth rate of the REBCO matrix compared with the BMO nanorods [15].

Fig. 2 Here

Figure 3(a) shows field dependences of  $J_c$  at 77 K in the Firework and the Aligned films under perpendicular magnetic fields. The Aligned film has a plateau of  $J_c$  at low magnetic fields and a shoulder around 2 T. This behavior of  $J_c$  is typically observed in the REBCO films with the  $c$ -axis correlated pinning centers [12], [13]. The shoulder is understood as the maximum field where the pinning center is saturated and effective so called as the matching field [12], [16], [17]. According to the nanorod density of the TEM images in Fig. 1, the matching field  $B_\Phi$  is calculated as 2 T, where it is calculated as  $B_\Phi = n^* \Phi_0$ , where  $\Phi_0$  is the quantized flux in superconductors as  $2.07 \times 10^{-15} \text{ Tm}^2$ . On the contrary, the Firework films have no shoulder possibly because of the not correlated pinning centers. We will discuss the reason why the Firework film has no shoulder later in this paper. In high fields over 4 T,  $J_c$  in the Firework film is larger than one in the Aligned film possibly due to the difference of  $T_c$  in the films.

Figure 3(b) shows the field angular dependence of  $J_c$  at 77 K and various magnetic fields of 1, 3, and 5 T. The angle is defined as angles from the  $c$ -axis of the films. The curves have peaks at  $0^\circ$  for each fields and samples. The peaks are apparently wide in the Firework films compared with the Aligned films. The angle distribution of the nanorod matches well with the peak width at low magnetic fields. This results support the correlation between the nanorods orientation and the anisotropy of  $J_c$ , especially about the peak structure. The peaks disappear at high magnetic fields and the similar curve appears for both the films. It indicates that the nanorods becomes less effective with increasing the field. This behavior is explained by the transition of the flux pinning state from the

single vortex pinning to the collective vortex pinning state where vortices support each other against the force from the flowing current [18]-[20]. It indicates that the peak width becomes wider with increasing the magnetic field at low fields because the contribution of the two kinds flux pinning changes. We note that the peak structure is tunable with the different nanorod distribution by changing the deposition conditions. For example, the Firework film has a very flat angular dependence at 77 K, 3 T. The effect of the nanorods remains even around or above the irreversible fields. These behaviors are understood as follows. The flux pinning in the nanorods doped the REBCO films divided into two components; the  $c$ -axis correlated pinning and the random pinning. The correlated pinning is caused by the nanorods, the twin boundaries, the grain boundaries, and the spiral dislocation. On the other hand, the random pinning is realized by the lattice dislocation, the oxygen vacancy, or other defects such as the precipitants. According to our transport results and the earlier studies, the correlated pinning is dominant at low magnetic field less than the matching field and the higher fields around or even above the irreversible fields. On the other hand, the random pinning is dominant at the mid fields. To confirm our speculation, we analyze the resistive properties to determine the flux pinning state.

Fig. 3 Here

Figure 4 shows field dependences of the activation energy and the critical exponents of the vortex liquid-glass transition. The activation energy  $U_0$  is the energy required to shake vortices off from the nanorods or other pinning centers. The inset figure in Fig. 4(a) shows logarithm of the resistivity  $\rho$  vs the inversed temperature. This curve is called as the Arrhenius plot which is generally used for the thermally activated phenomena. The flux flow just above  $T_c$  is dominated by the thermally activated flux flow [1]-[4]. Therefore,  $U_0$  is estimated from the linear fitting results as shown in the inset figure of Fig. 4(a). As results, field dependences of  $U_0$  are shown as the main panel of the Fig. 4(a).  $U_0$  monotonically decreases with increasing magnetic fields. They have a shoulder structure near  $B_\Phi \sim 2 \text{ T}$ . This is unique behavior of  $U_0$  with a plateau structure and the sharp shoulder in the Aligned films. This indicates that the single vortex pinning state is realized in the field range below the shoulder field because  $U_0$  should be held at the same value in the single vortex pinning state [19], [21]. This matching behavior in  $U_0$  is more apparent than in  $J_c$  with the smooth shoulder around the matching field. On the other hand in the Fireworks film, a vague transition from the plateau to the slope appears. It indicates the distribution of the nanorod density in the films as show in Fig. 1(d). In higher magnetic field,  $U_0$  is proportional to  $B^{-1}$  in each film. This behavior is an evidence of the collective vortex pinning [19], [21]. These results suggest that a part of vortices are located at the interstitial area out of the nanorods and the motion of the interstitial vortices or the bundle motion of vortices are the key to determine the flux pinning force above  $B_\Phi$  in the Aligned films or above small fields in the Firework film.

Fig. 4 Here

The critical exponent of the liquid-glass state transition of vortices  $s$  is estimated from the linear fitting for the inverse of the temperature derivative of the logarithm resistivity vs temperature near the glass temperature [3], [22]-[26]. The fitting is shown in the inset figure in Fig.4(b). From the fitting results, in addition to  $s$ , the glass-liquid transition temperature of the vortex  $T_g$  is calculated as a extrapolated temperature. The inset figure is a typical fitting result for the Aligned film. Adopting the same fitting method for the other conditions and the Firework film, the field dependences of  $s$  in both films are shown in the main panel in Fig. 4(b). Among the all field,  $s$  in the Firework film is larger than the one in the Aligned film.  $s$  in the Firework is almost constant near 8, on the other hand,  $s$  in the Aligned film monotonically increases for larger fields and is saturated at around 1 T as a half of  $B_\phi$ . The small  $s$  value as 4-6 at low fields indicates the existence of the Bose glass state where the vortices are individually trapped in the correlated pinning centers [19], [22], [23]. In the Bose glass state, the flux pinning by the nanorods becomes strong and the ideal matching effect is expected to appear.

Figure 5 show normalized  $F_p$  curves as a function of normalized magnetic fields in each sample. At low temperatures,  $F_p$  is normalized by its maximum value. The magnetic field is also normalized by the peak field  $B_{\text{peak}}$  where  $F_p$  becomes the maximum. On the other hand at high temperatures, two peaks appear possibly due to the many rods pinning or the delocalization of the vortex [19], [27], [28], [33]. The peak fields  $B_{\text{peak}}$  are estimated with 3 order polynomial fittings even when the two peaks of  $F_p$  appear. Finally,  $F_p$  is normalized at the higher peak fields  $B_{\text{peak}}$  at each temperature. As results, the  $F_p$  curves in the Aligned film have the linear field dependences at the field less than the magnetic field at low temperature less than 80 K. This behavior is explained by the individual flux pinning by the nanorods in the Aligned film and  $F_p$  should be proportional to  $B$ . On the other hand, the  $F_p$  curves in the Firework films have round-shaped at all temperature. To speculate the origin of the flux pinning in the Firework film, the typical  $F_p$  curve is plotted as a dashed line according to the collective pinning state with an equation of  $F_p \propto (B/B_{\text{irr}})^{-0.5}(1-B/B_{\text{irr}})^2$  by randomly distributed pinning centers [21], [29]. It is clear that the  $F_p$  curves at the low temperatures in the Firework film matches well with the typical curve. According to the results, the flux pinning state in the films are quite different; the individual vortex pinning state with the Bose glass state at less than  $B_\phi$  in the Aligned films, and the collective vortex pinning state over wide temperature range in the Firework film [30], [31].

Fig. 5 Here

To summarize the transport measurements, phase diagrams as functions of magnetic field and temperature are plotted with the irreversible line [32], the glass-liquid transition line, and the peak field of  $F_p$  as shown in Fig. 6. As explained previously,

the irreversible magnetic field  $B_{\text{irr}}$  is defined as the  $J_c$  criterion of  $10 \text{ A/cm}^2$ . The glass transition field  $B_g$  is estimated based the extrapolation of the critical exponent fitting. The peak field of  $F_p$  is estimated from the polynomial fitting of the field dependences of peak  $F_p$ . It is apparent that the peak field  $B_{\text{peak}}$  has a plateau at the low temperatures due to the matching effect where the flux pinning by the  $c$ -axis correlated nanorods becomes most dominant in the Aligned films. On the other hand,  $B_{\text{peak}}$  in the Firework films monotonically increases with decreasing the temperature. This behavior is observed in the films with the randomly distributed pinning centers such as the pure REBCO films, where  $B_{\text{peak}}$  is scaled by  $B_{\text{irr}}$ . According to the systematic experimental results on the flux pinning properties with different nanorod structures, we confirmed a strong relationship between the shape of the nanorods and the flux pinning landscapes. With the correlated pinning centers such as the straight nanorods in Fig. 1(b), the peak field  $B_{\text{peak}}$  is saturated at the low temperature, and the Bose glass state would be realized. On the contrary with the randomly distributed or the non-correlated nanorods, the peak field  $B_{\text{peak}}$  monotonically increases at low temperatures because  $B_{\text{peak}}$  is scaled with the irreversible field  $B_{\text{irr}}$ . Based on this analysis, the microstructure of the nanorods in the films would be speculated without TEM imaging.

Next, the high temperature range is shown in Fig. 6(b) with the normalized temperature with  $T_c$ . The Aligned film has a shoulder in the irreversibility line and the glass-liquid transition line near the matching field of 2 T. On the other hand, the Firework films has a smooth curve for both the lines. The irreversibility line should show shoulders in the Bose glass phase [12], [32], [33].

Fig.6 Here

#### IV. CONCLUSION

The influence of the nanorod microstructure on the flux pinning state in BHO doped SmBCO films is investigated using high field magnets with the rotator stage. The inclined nanorods are realized using the low temperature growth technique. The alignment of the nanorods are controlled with energy of the laser pulses. The films with aligned nanorods show the strong flux pinning along the  $c$ -axis, which disappears at high magnetic fields. This behavior is supported by the existence of the Bose glass phase in the film according to the discussion on the activation energy and the critical exponent measured by the transport properties. On the contrary, the films with firework nanorods show weak correlated pinning along  $c$ -axis such as the disappearance of the Bose glass state or the weak matching effect. Moreover, the temperature dependence of the peak field of the flux pinning force is explained by the flux pinning model for the correlated nanorods and the random pinning model for the films with the aligned and the firework nanorods structures. We confirmed the strong relationship between the microstructure of the BMO nanorods and the flux pinning landscapes according to the transport properties in the BMO doped REBCO films.

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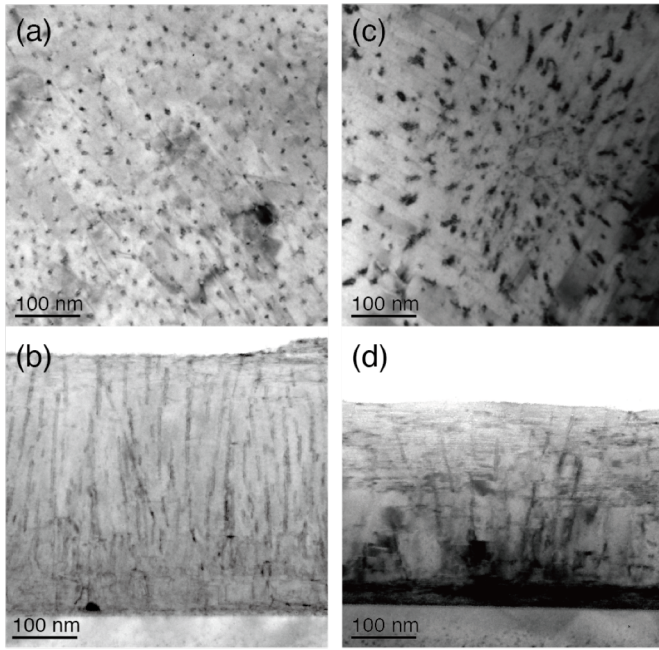


Fig. 1. TEM images for plan-view and elevation-view in (a), (b) the Aligned and (c), (d) the Firework films.

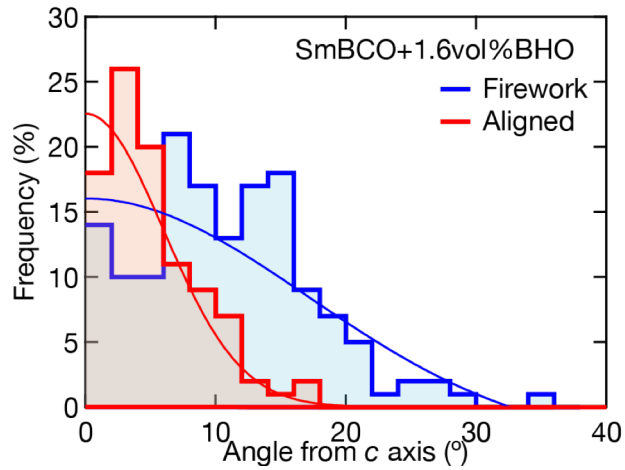


Fig. 2. Population histogram of the nanorod angle from the  $c$ -axis of the SmBCO matrix in the Aligned and Firework films. The thin lines are Gaussian fitting lines for both the films.

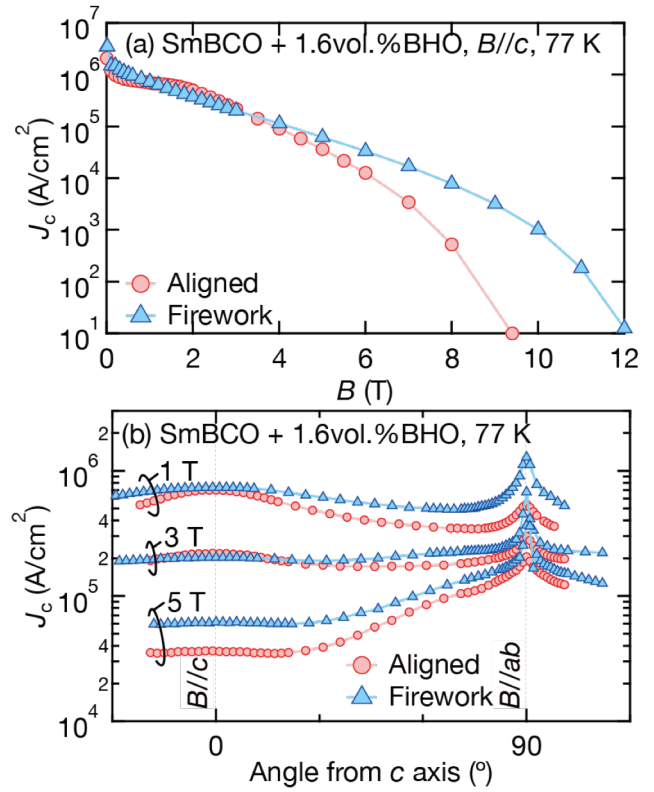


Fig. 3. (a) Field dependences of the critical current density  $J_c$  in the Aligned and the Firework films with the perpendicular field at 77 K. (b) Field angular dependences of  $J_c$  in each film at 77 K, 1, 3, and 5 T.

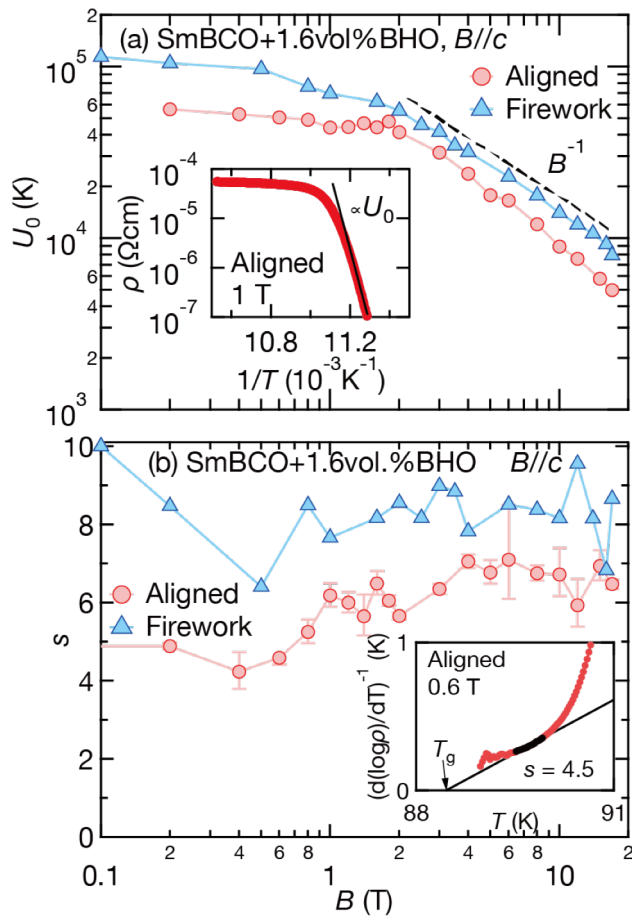


Fig. 4. Field dependences of (a) the activation energy  $U_0$  of the vortex at thermally activated flux flow state just above  $T_c$  and (b) the critical exponent  $s$  close to the liquid glass transition temperatures in the Aligned and the Firework films. Inset shows the typical fitting results to estimate the activation energy and the critical exponent.

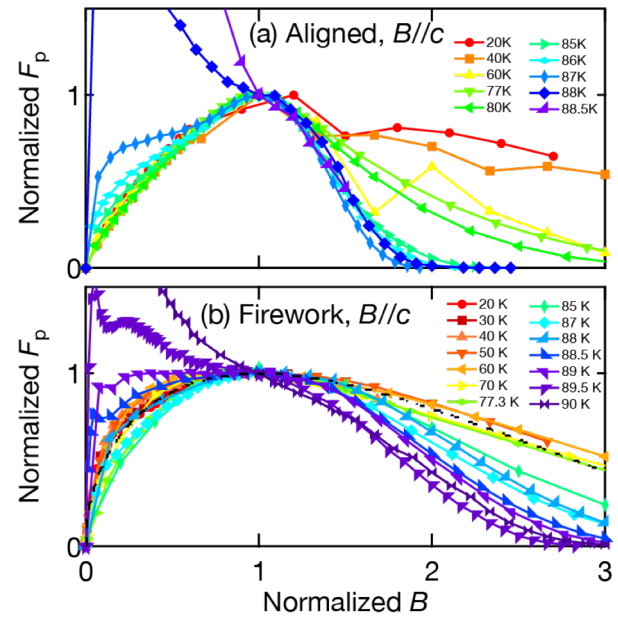


Fig. 5. Normalized magnetic field dependences of the normalized flux pinning force density  $F_p$  in (a) Aligned and (b) Firework films at various temperatures with the perpendicular magnetic field. Typical fitting results for the REBCO films are plotted as the dotted line.

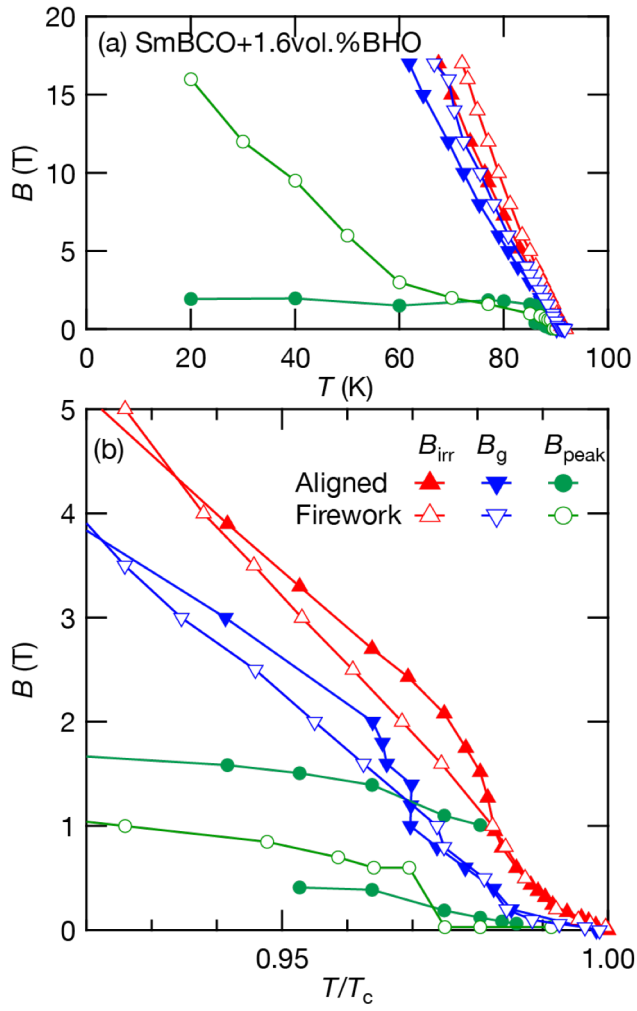


Fig. 6. Temperature dependences of the irreversible field  $B_{irr}$  (up triangle), the glass transition temperature  $T_g$  (down triangle), and the peak field of the flux pinning force density  $B_{peak}$  (circles) in the Aligned (filled) and the Firework (blank) films.